

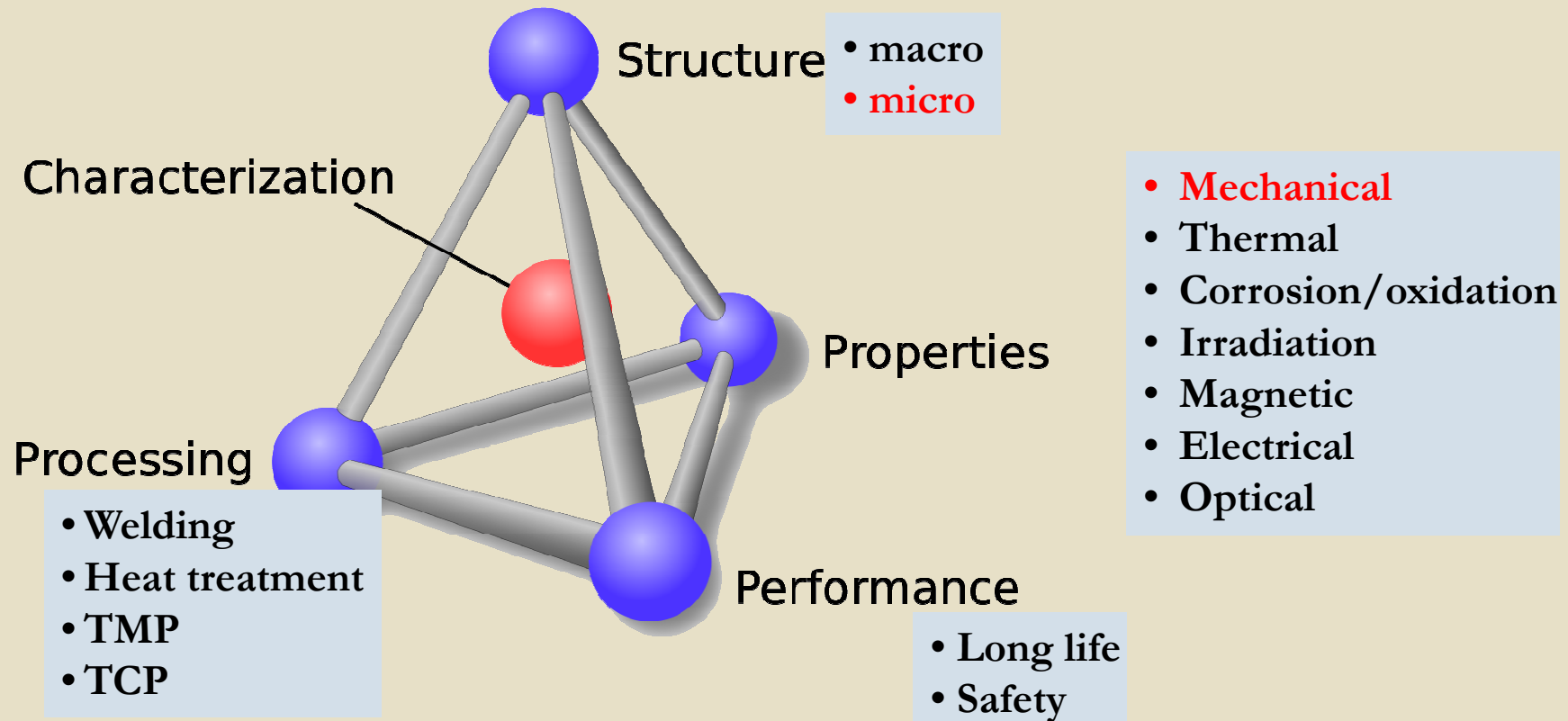


# Characterization of Engineering Materials

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# Need for characterization



- Understanding materials behaviour
- Evaluation of products
- Life assessment and prediction
- Development of materials and processes

# Materials Environment

Corrosive

High Temperature

Compressive stress

Residual stress

Tensile stress

Fatigue

Ratchetting

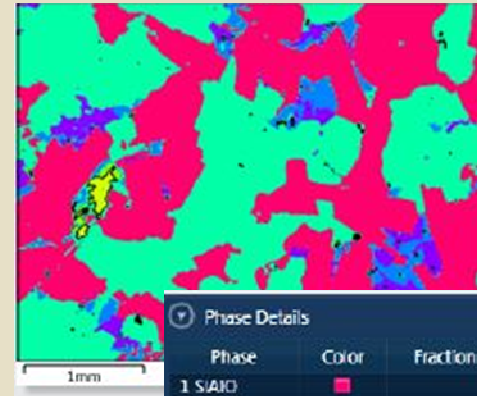


Deformation



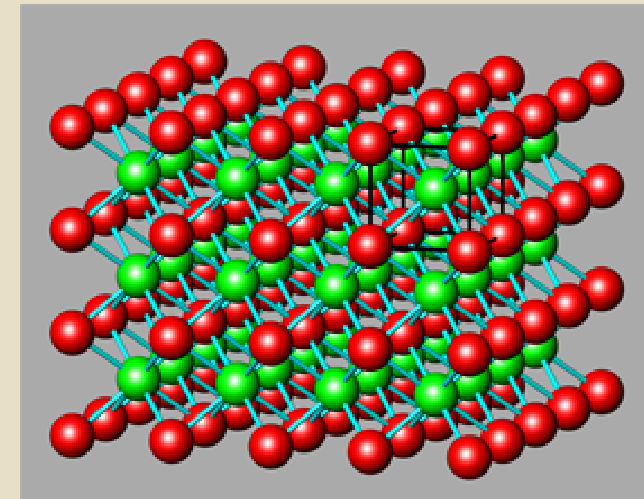
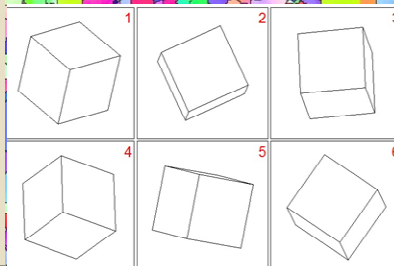
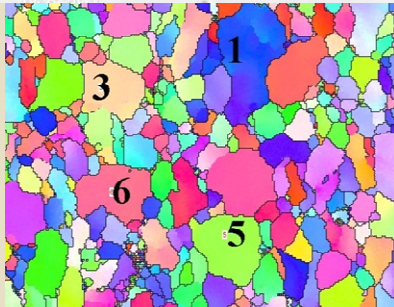
Microstructural defects

# What is Microstructure?



| Phase Details |             |              |          |
|---------------|-------------|--------------|----------|
| Phase         | Color       | Fraction (%) | Pixel... |
| 1 SiAlO       | Red         | 45.7         | 89,799   |
| 2 SiFeO       | Green       | 44.9         | 88,188   |
| 3 SiO         | Blue        | 3.4          | 6,708    |
| 4 SiKO        | Purple      | 3.7          | 7,372    |
| 5 FeTiO       | Yellow      | 0.3          | 685      |
| 6 FeO         | Light Green | 0.2          | 445      |
| 7 CaPO        | Dark Blue   | 0.0          | 55       |

- Morphology
- Microchemistry/  
Composition
- Crystal structure
- Texture

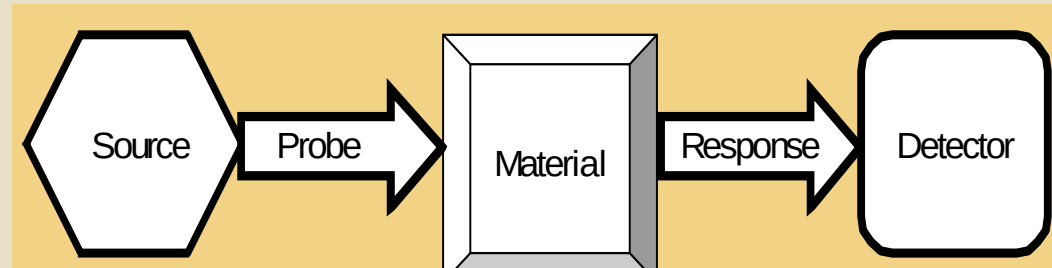




# How do we characterize?

## *Nature of materials*

- Crystalline, amorphous
- Metal, Semi-conductor, Ceramic/insulator
- Bulk or Surface



## *Techniques*

### Spectroscopy

- ❖ Optical  
FTIR, Raman, ...
- ❖ Mass
  - ❖ SIMS, ICPMS,...
- ❖ X-ray  
XRF, XPS, ...
- ❖ Nuclear  
NMR, Mossbauer,...

### Diffraction

- ⇒ X-ray
- ⇒ Electron
- ⇒ Neutron

### Microscopy

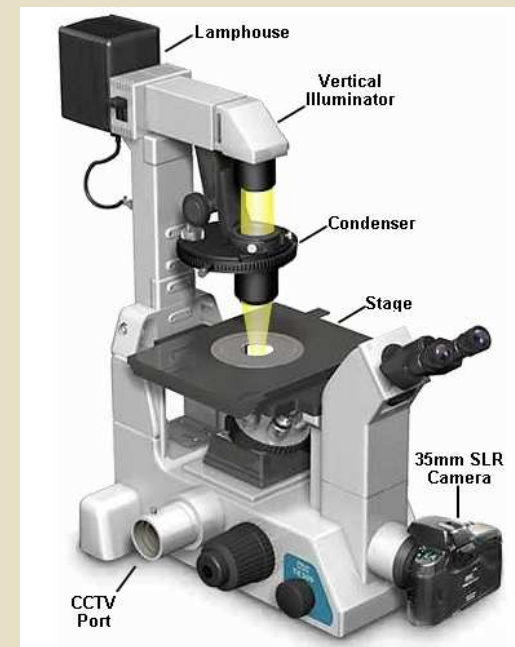
- Optical
- Electron  
SEM, TEM
- Scanning probe
  - Tunneling
  - AFM

# Optical Microscopy

- Resolution of human eyes is of the order of 0.3 mm.
- Features to be observed:
  - Grain shapes and size: from  $<\mu\text{m}$  to the cm regime
  - Precipitate size: mostly in the  $\mu\text{m}$  regime
  - Volume fractions and distributions of various phases
  - Defects such as cracks and voids:  $<\mu\text{m}$  to the cm regime

## Compound Optical Microscope

- Visible light as illumination source
- Range of samples characterized - almost unlimited for solids
- Usually nondestructive; sample preparation may involve material removal



# Optical Microscopy- Resolution

- **Transmitted OM** - transparent specimens like thin section of rocks, minerals and single crystals- *absorption*
- **Reflected OM** - opaque specimens like metals, ceramics, semiconductors - *reflection*

Inverted microscope  
- Materials science

**Typical magnification in optical microscope is up to 1000x**

## Resolution and Magnification:

**d** of two points is such that **central maximum** of one source falls on the **first diffraction minimum** of the other

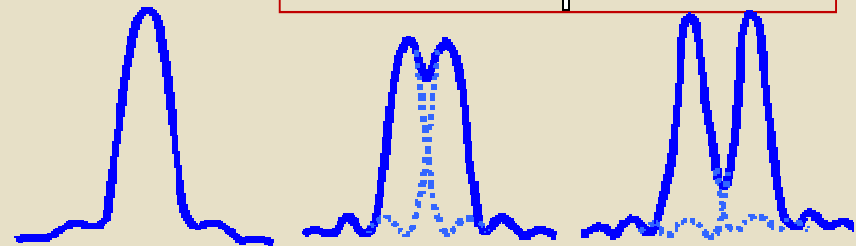
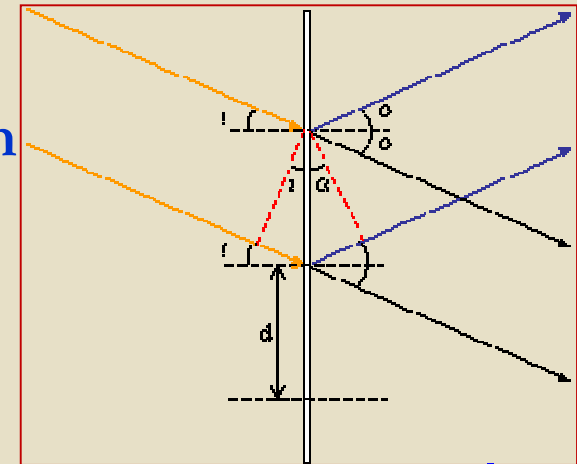
$$d = \lambda / 2n \sin \alpha$$

$$d \sim 0.3 \mu\text{m} \text{ for } \lambda = 0.55 \mu\text{m}$$

Useful magnification

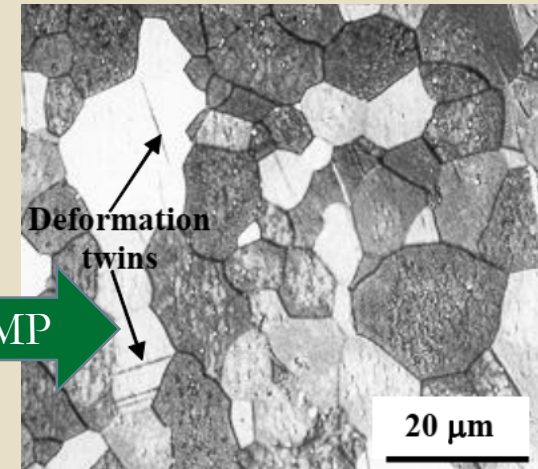
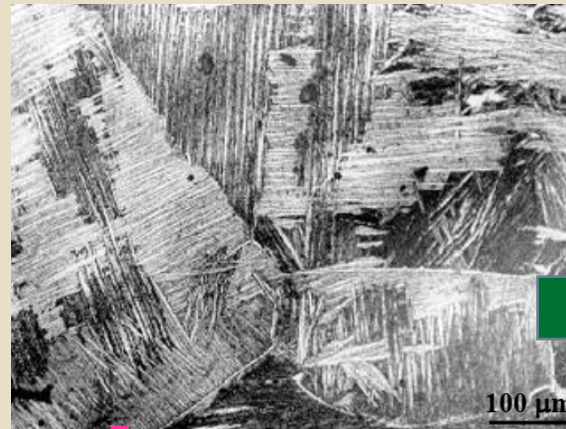
$$= 0.3 \mu\text{m} / 0.3 \text{mm} = 1000$$

Any further zooming is of no use!



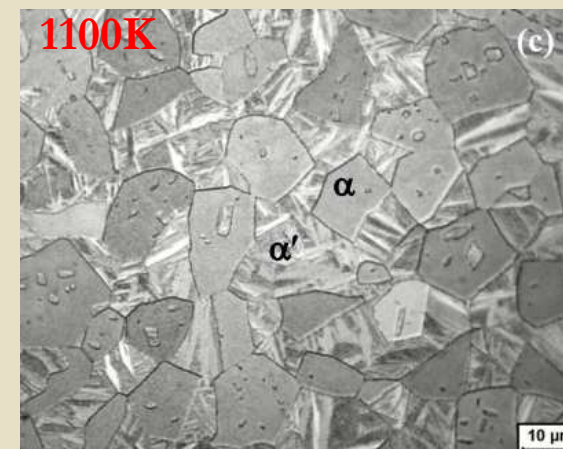
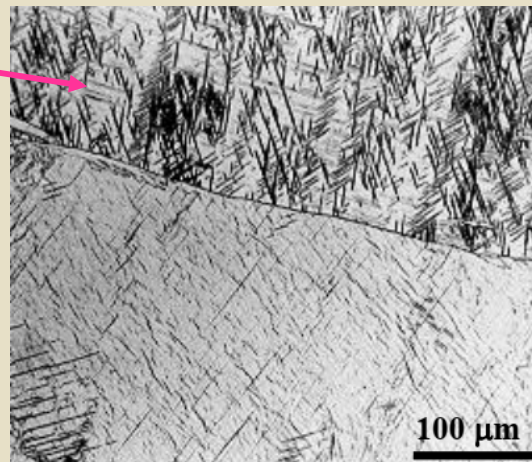
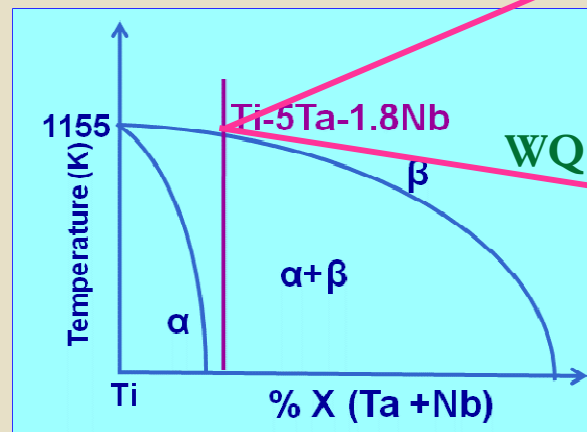
# Morphology

- Phase transformation
- Processing Route
- Volume fraction & size of phases



TMP

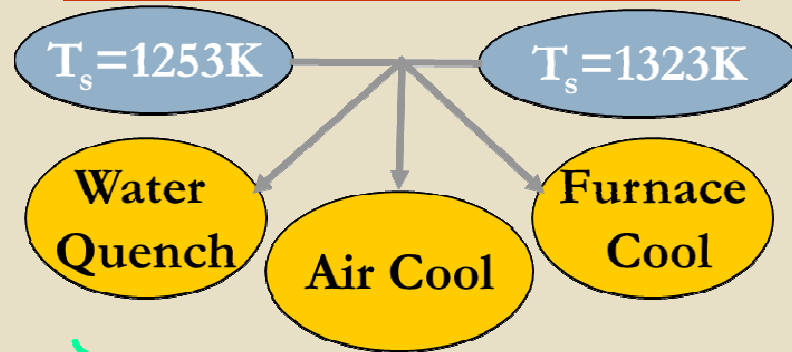
FC



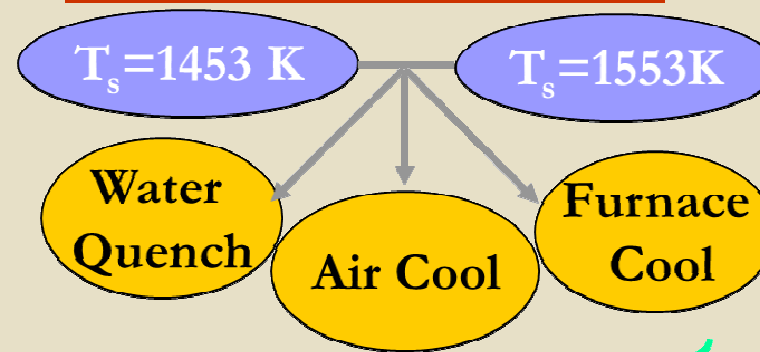


# Identification of phase regimes- 9Cr-1W-0.06Ta RAFM steel

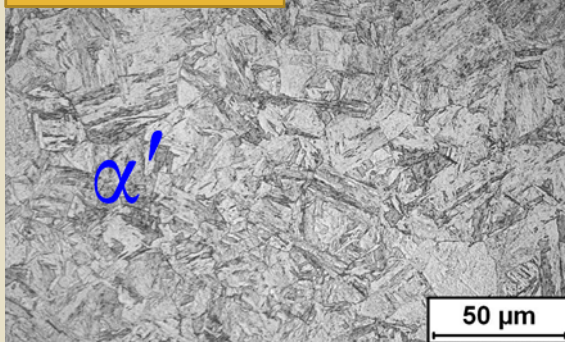
## Effect of cooling rate



## Effect of $T_s$

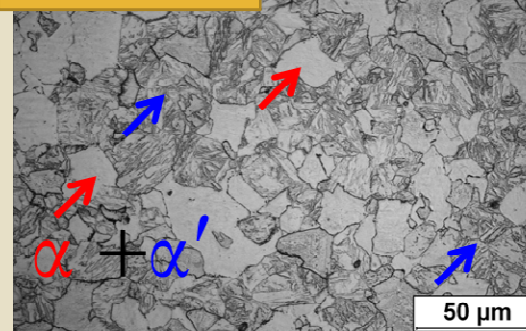


1253K / AC

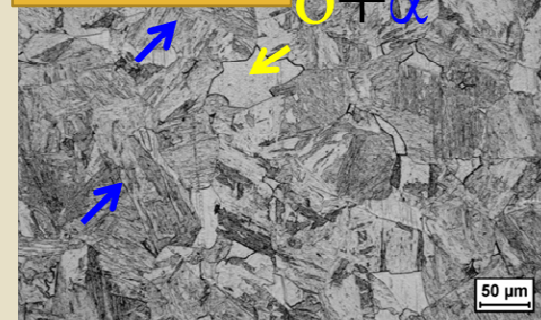


1253K >  $\gamma$  phase field < 1453K

1253K / FC

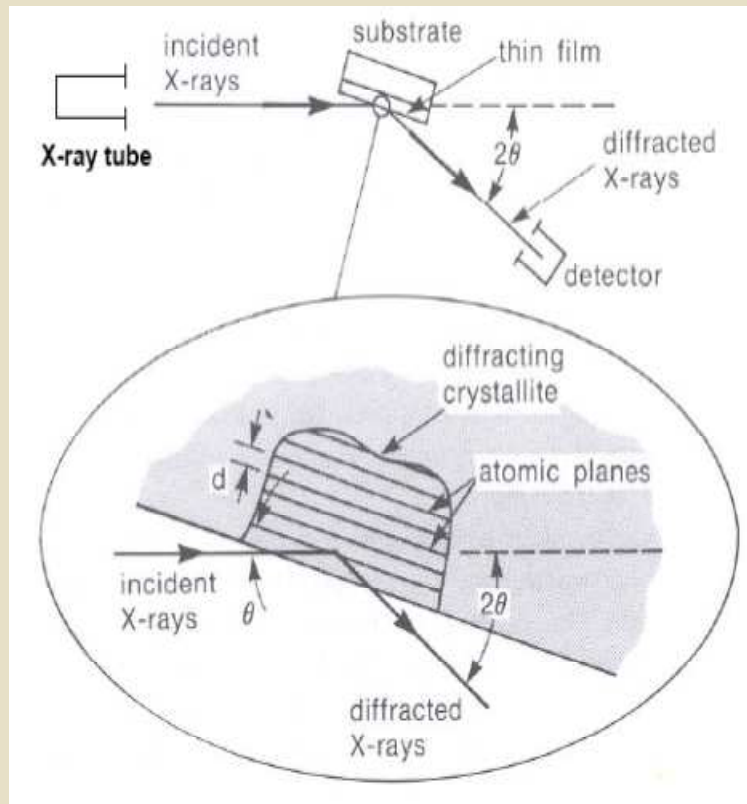


1553K / AC



1453K >  $T_\delta$  < 1553K

# X-ray Diffraction



Bragg's law:  $2d_{hkl} \sin \theta = n\lambda$

Source:

Monochromatic – Powder XRD

White- Single crystals

|                 |           |
|-----------------|-----------|
| Mo K $\alpha$ 1 | 0.709300Å |
| Cu K $\alpha$ 1 | 1.54056Å  |
| Co K $\alpha$ 1 | 1.788965Å |
| Cr K $\alpha$ 1 | 2.28970Å  |

**Sample: Powder or bulk specimen with flat and smooth surface**

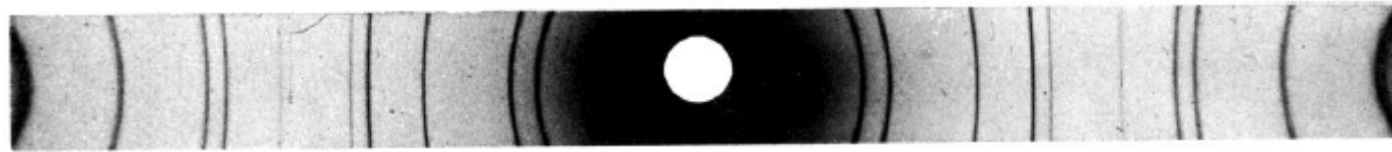
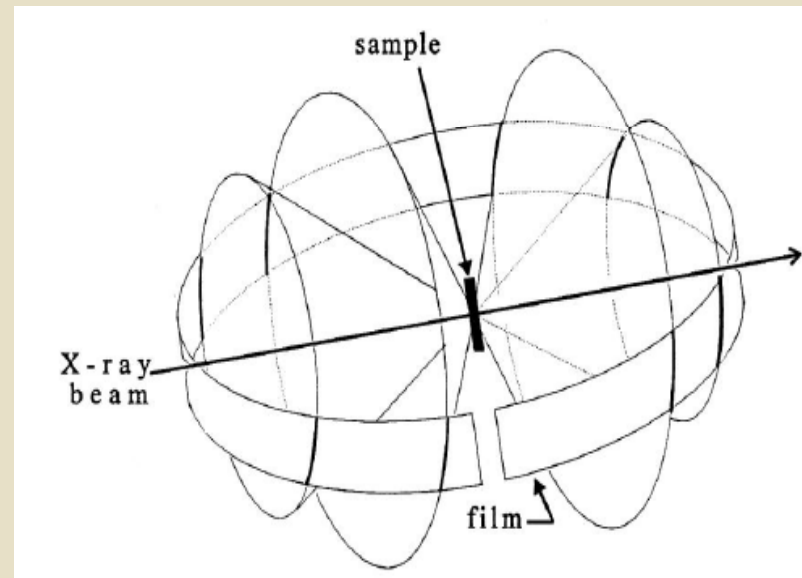
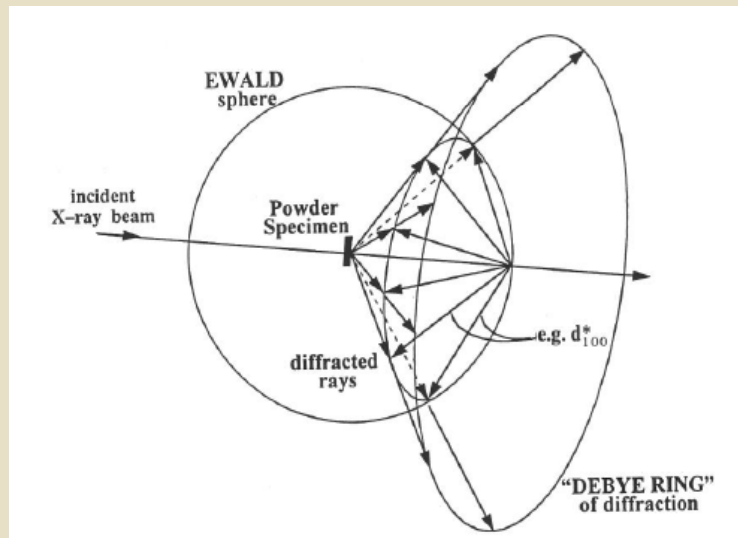
**Geometry:**

**$\theta:2\theta$**  : tube is fixed, sample rotates at  $\theta^\circ/\text{min}$  and detector rotates at  $2\theta^\circ/\text{min}$ .

**$\theta:\theta$**  : sample is fixed and tube rotates at a rate of  $\theta^\circ/\text{min}$  and detector rotates at a rate of  $\theta^\circ/\text{min}$ .

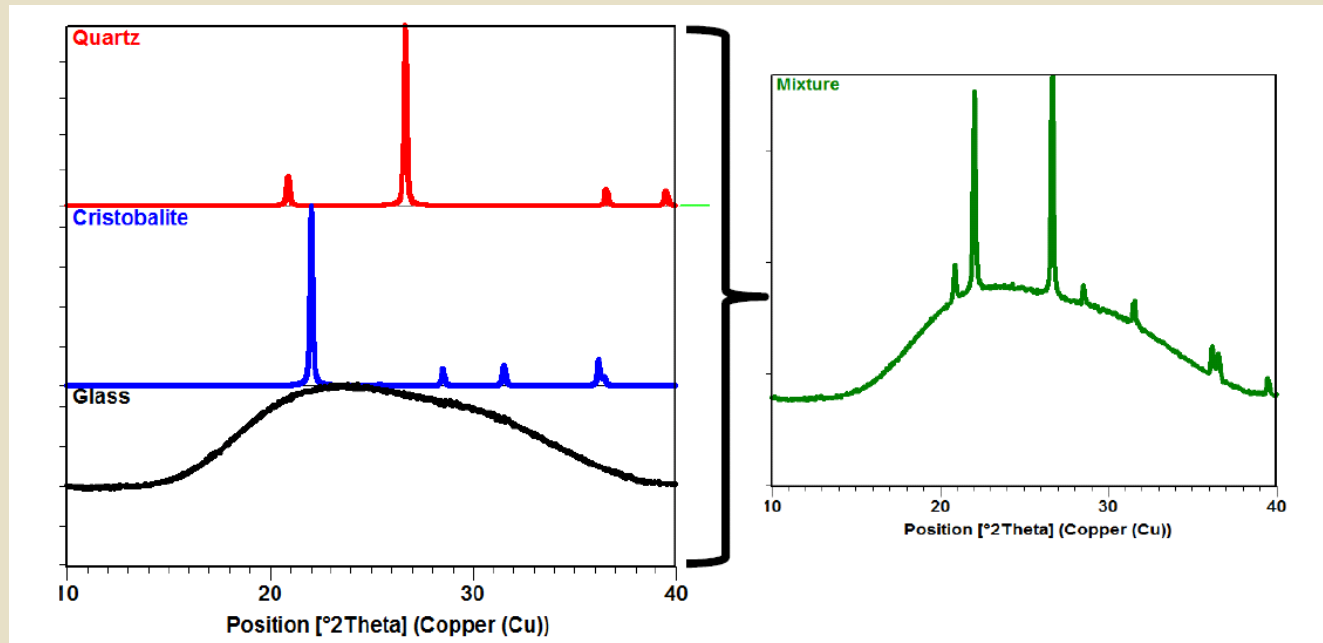
# Powder X-ray Diffraction

- Powder- collection of small crystallites oriented randomly
- X-ray beam falls simultaneously on many crystals and sees all the planes
- By varying the angle all the possible diffraction peaks detected



Au, FCC

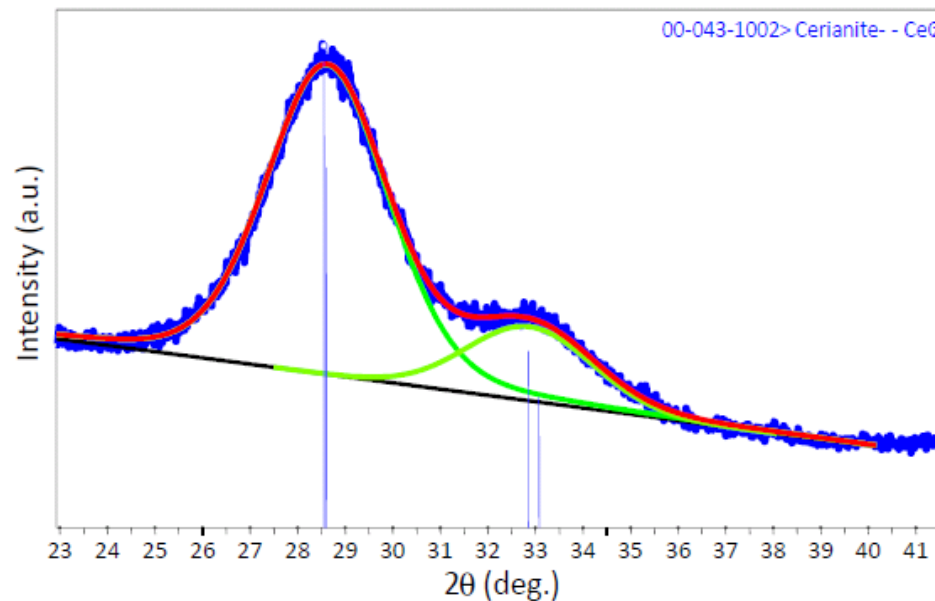
# Powder X-ray Diffraction



- |                        |                                    |
|------------------------|------------------------------------|
| ■ peak position        | ➤ dimension of the elementary cell |
| ■ peak intensity       | ➤ content of the elementary cell   |
| ■ peak broadening      | ➤ strain/crystallite size          |
| ■ scaling factor       | ➤ quantitative phase amount        |
| ■ diffuse background   | ➤ false order                      |
| ■ modulated background | ➤ close order                      |

# Crystallite size and microstrain

- Crystallites smaller than ~120nm create broadening of diffraction peaks
  - this peak broadening can be used to quantify the average crystallite size of nanoparticles using the Scherrer equation
  - must know the contribution of peak width from the instrument by using a calibration curve
- microstrain may also create peak broadening
  - analyzing the peak widths over a long range of 2theta using a Williamson-Hull plot can let you separate microstrain and crystallite size



$$B(2\theta) = \frac{K\lambda}{L \cos \theta}$$



# Texture analysis

- Orientation of Crystals is not random
- Preferred Orientation of crystals



Anisotropy of properties

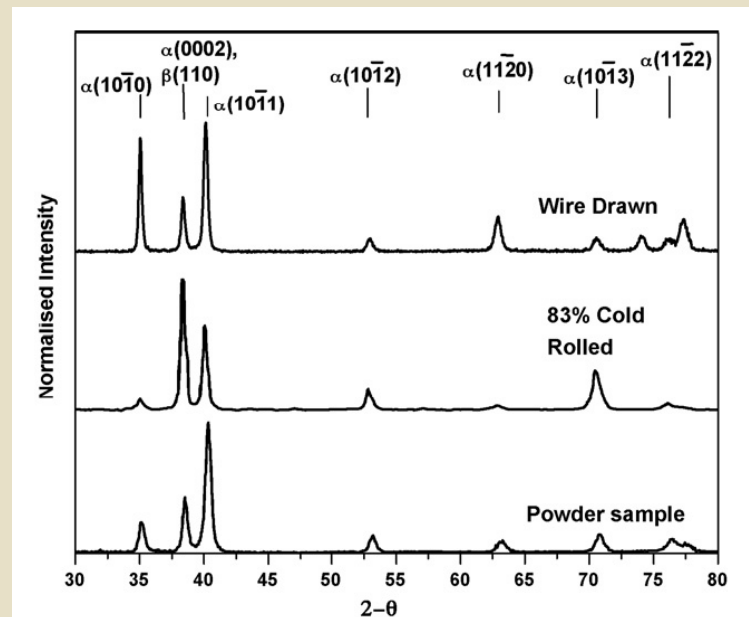
## Types

- Deformation
- Recrystallisation
- Transformation

Macro texture: XRD

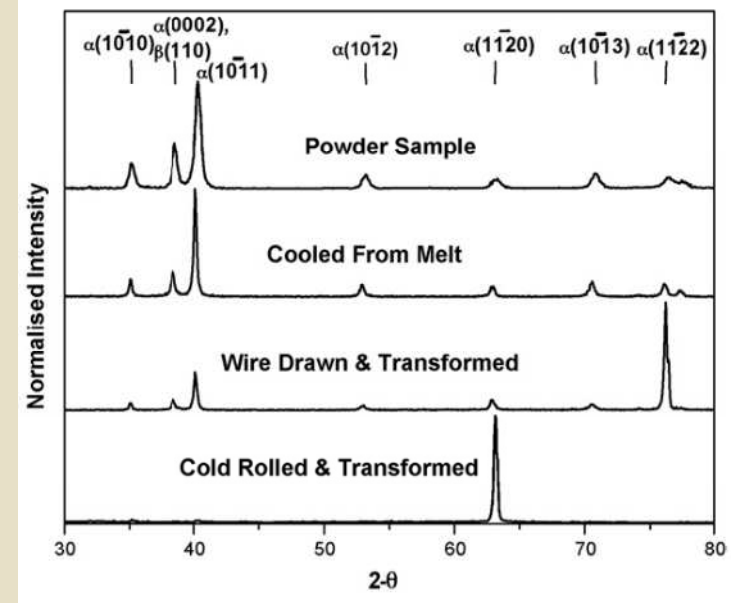
Micro texture : EBSD in SEM, OIM in TEM

## Deformation texture

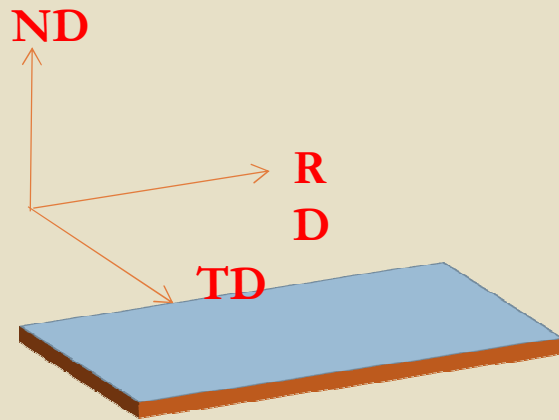


Ti-5Ta-  
2Nb  
alloy

## Transformation texture

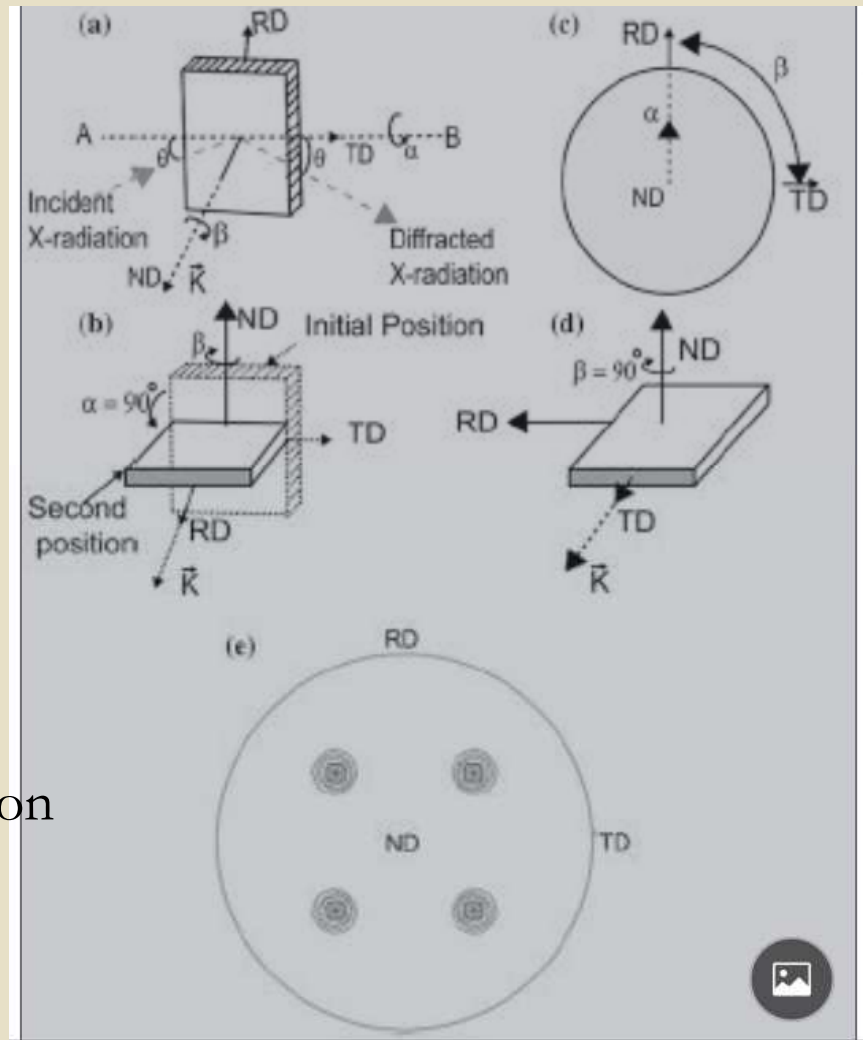


# Texture Representation



## Pole figures

Stereographic projection of distribution of crystallographic planes w.r.t the specimen co-ordinates



# Electron Microscopy techniques for microstructural analysis

- **Optical**
  - **Electron**
  - **Ion**
- ❖ Negative Charge - Beam Focus / Deflection by Electric Coils
  - ❖ Low Mass – Non destructive for most materials
  - ❖ Dual Behavior of Electron – Wave & Particle

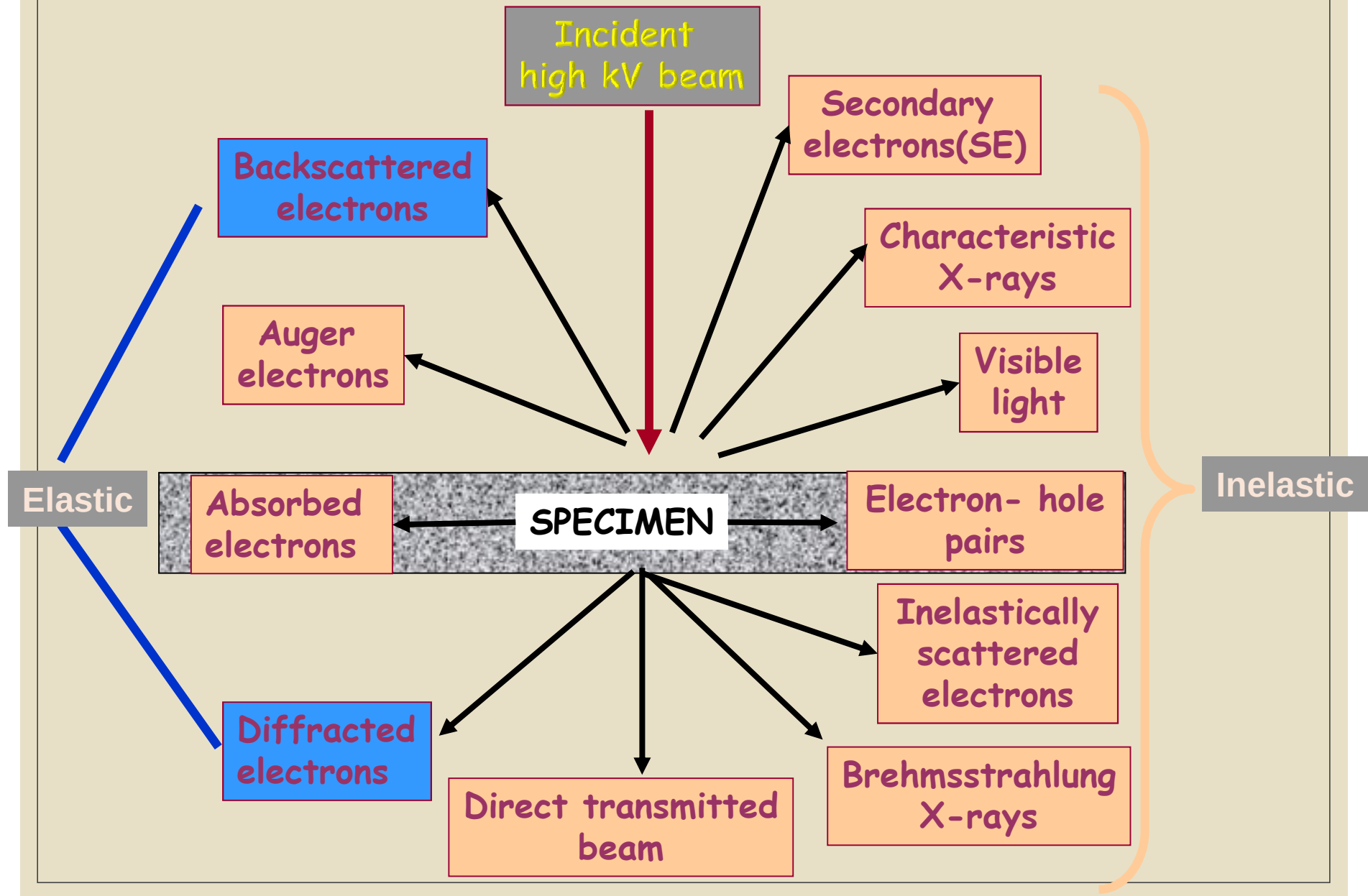
## Wave Behavior

- ❖ Elastic Scattering - Images (phase / amplitude contrast)
- ❖ Diffraction patterns – Crystallography
- ❖ Shorter Wavelength– Better Resolution

## Particle Behavior

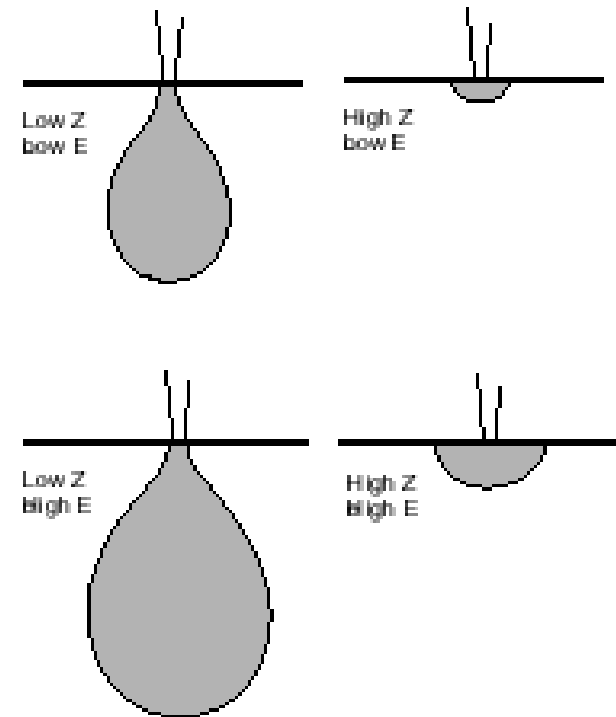
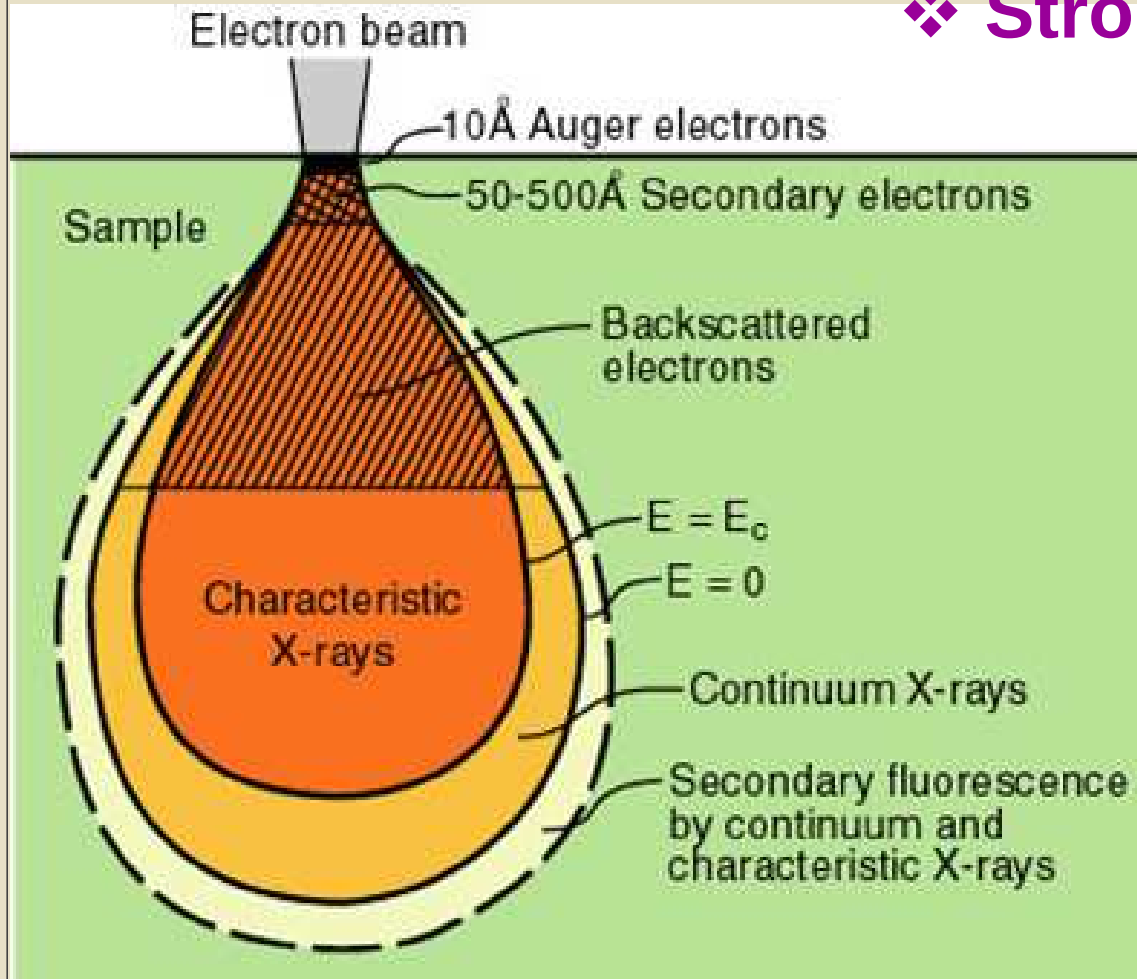
- ❖ Electron specimen interactions
- ❖ No. of signals generated - Inelastic scattering, Ionisation
- ❖ Wealth of information - Chemical information

# ELECTRON-MATTER INTERACTION



# Volume of Excitation

❖ Strong function of E, Z



Surface analytical  
and bulk techniques

## Spatial Resolution

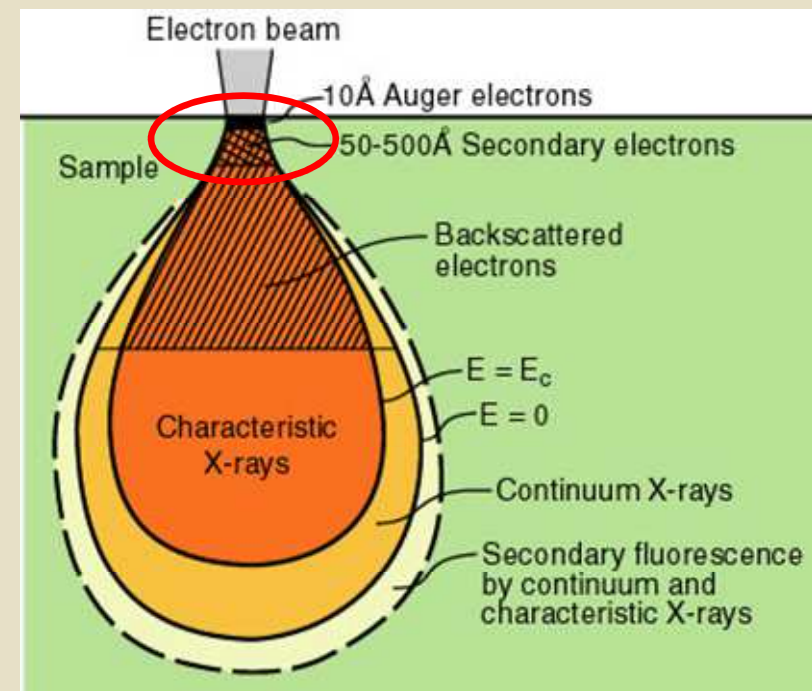
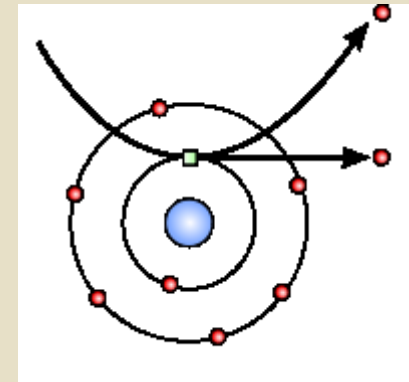


# Comparing SEM and TEM

|                                            | TEM                                                                                                       | SEM                                                                      |
|--------------------------------------------|-----------------------------------------------------------------------------------------------------------|--------------------------------------------------------------------------|
| <i>Electron Beam</i> ▶                     | Broad, static beams                                                                                       | Beam focused to fine point; sample is scanned line by line               |
| <i>Voltages Needed</i> ▶                   | TEM voltage ranges from 60-300,000 volts                                                                  | Accelerating voltage much lower; not necessary to penetrate the specimen |
| <i>Interaction of the beam electrons</i> ▶ | Specimen must be very thin                                                                                | Wide range of specimens allowed; simplifies sample preparation           |
| <i>Imaging</i> ▶                           | Electrons must pass through and be transmitted by the specimen                                            | Information needed is collected near the surface of the specimen         |
| <i>Image Rendering</i> ▶                   | Transmitted electrons are collectively focused by the objective lens and magnified to create a real image | Beam is scanned along the surface of the sample to build up the image    |

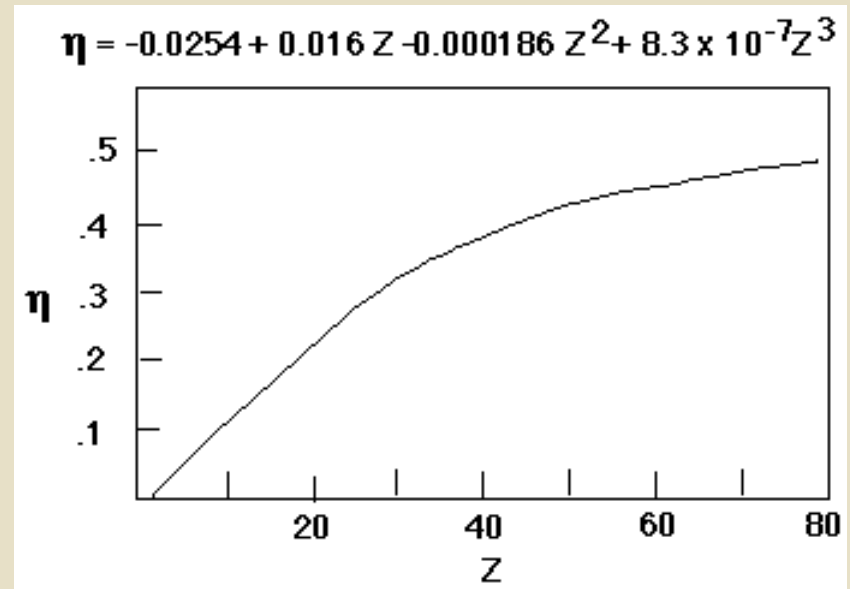
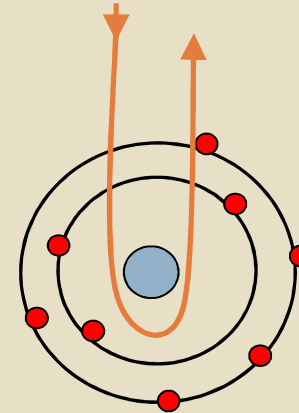
# Secondary electrons (SE)

- Generated from the collision between the incoming electrons and the loosely bonded outer electrons
- Low energy electrons ( $\sim 10\text{-}50\text{ eV}$ )
- Only SE generated close to surface escape (topographic information is obtained)
- Number of SE is greater than the number of incoming electrons

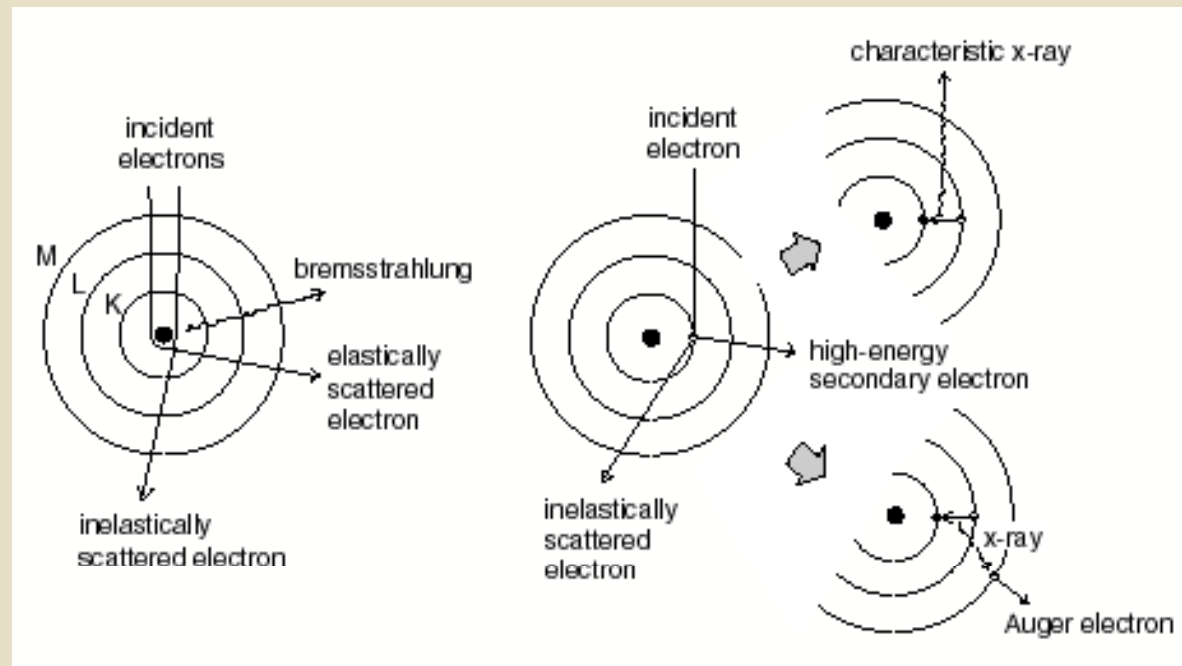


# Backscattered electrons (BSE)

- ❑ A fraction of the incident electrons retarded by the electro-magnetic field of the nucleus and if the scattering angle  $> 180^\circ$  the electron can escape from the surface
- ❑ High energy electrons (elastic scattering)
- ❑ Fewer BSE than  $E_e \approx E_0$
- ❑ Probes higher depth than SE
- ❑ Depends of atomic number
- ❑ Undergoes Diffraction- EBSD for Crystallographic identity & microtexture



# X-ray emission - Principle



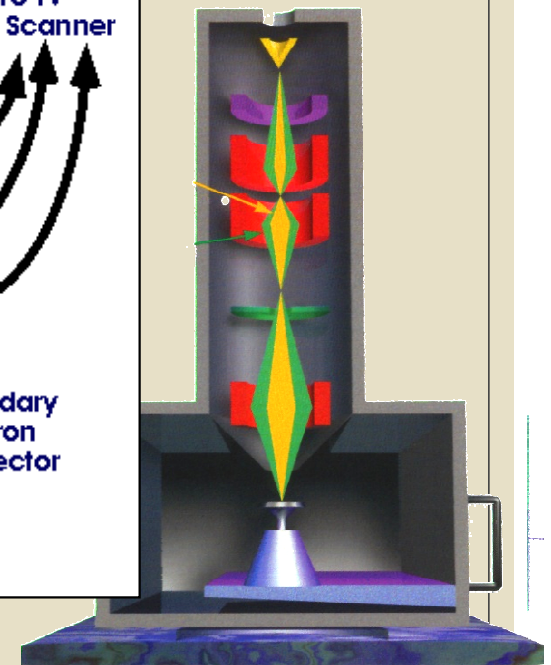
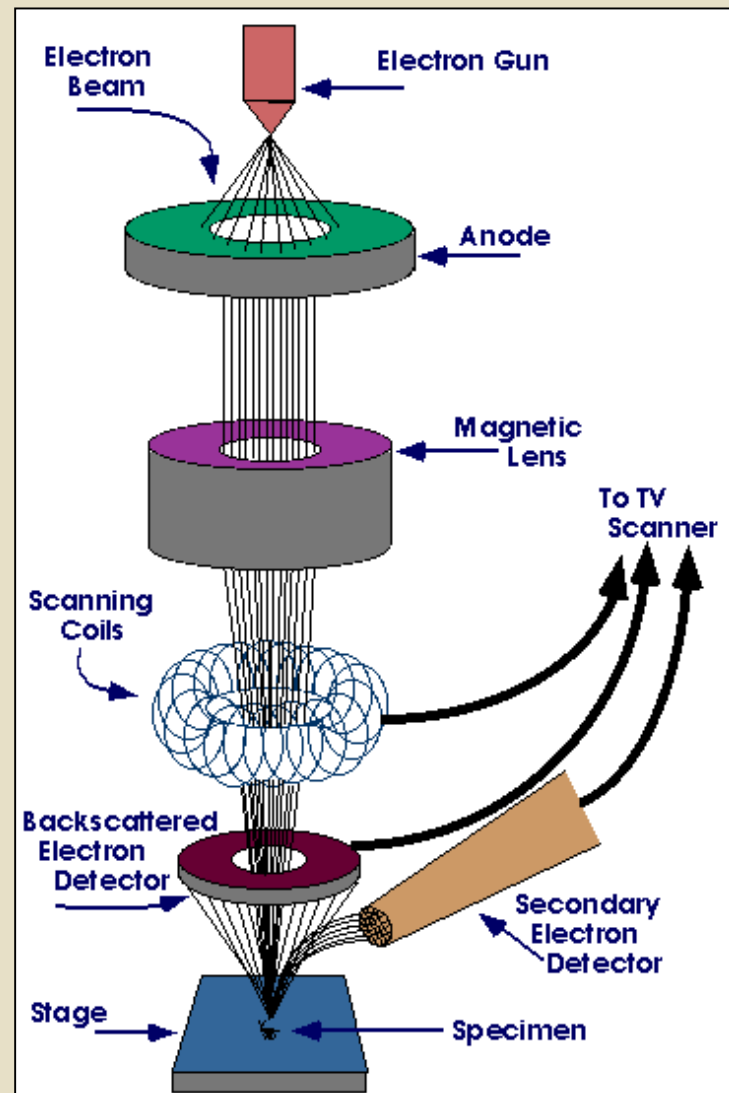
**Inner-shell ionization => K-shell or higher-shell vacancies or heat depending on Critical ionization or excitation potential  $E_c$**

**De-excitation => Characteristic X-radiation or Auger electron**

- X-rays emitted from the atom have a characteristic energy unique to the element from which it originated.
- These signals are collected and sorted according to energy or wavelength to yield elemental composition from micron size features of the specimens.

# Scanning Electron Microscope

- Gun
  - Filament
  - Wehnelt
  - Anode
- Electromagnetic Lenses
  - Usually three
- Spray apertures
  - At least one. More with higher resolution
- Scanning coils
  - Two sets X and Y deflection
- Stigmators
  - 1-2 sets
- Detectors
  - Could be several

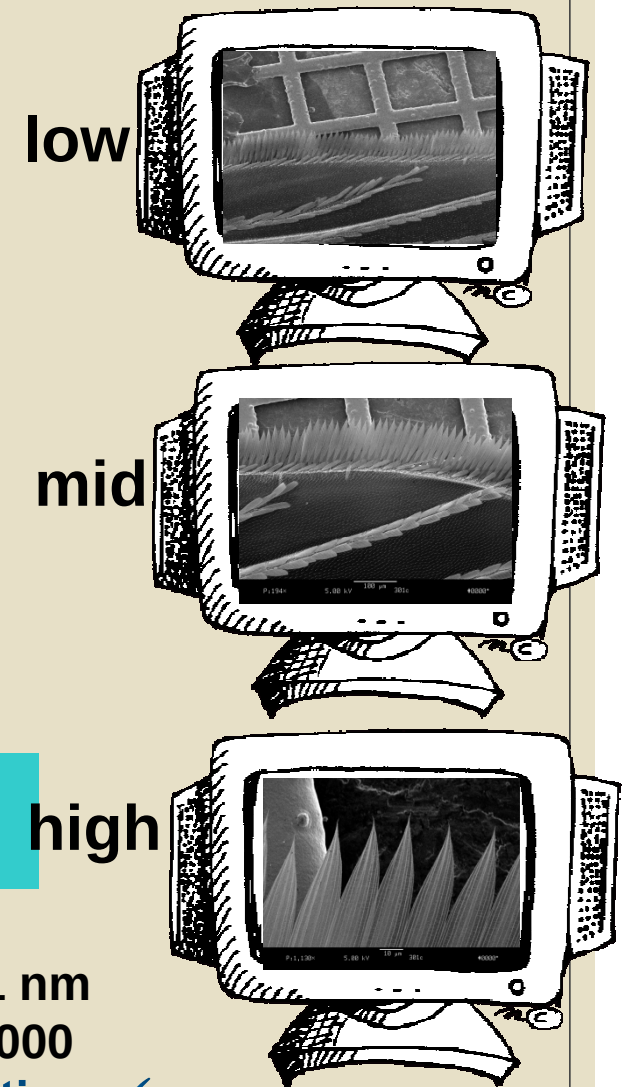
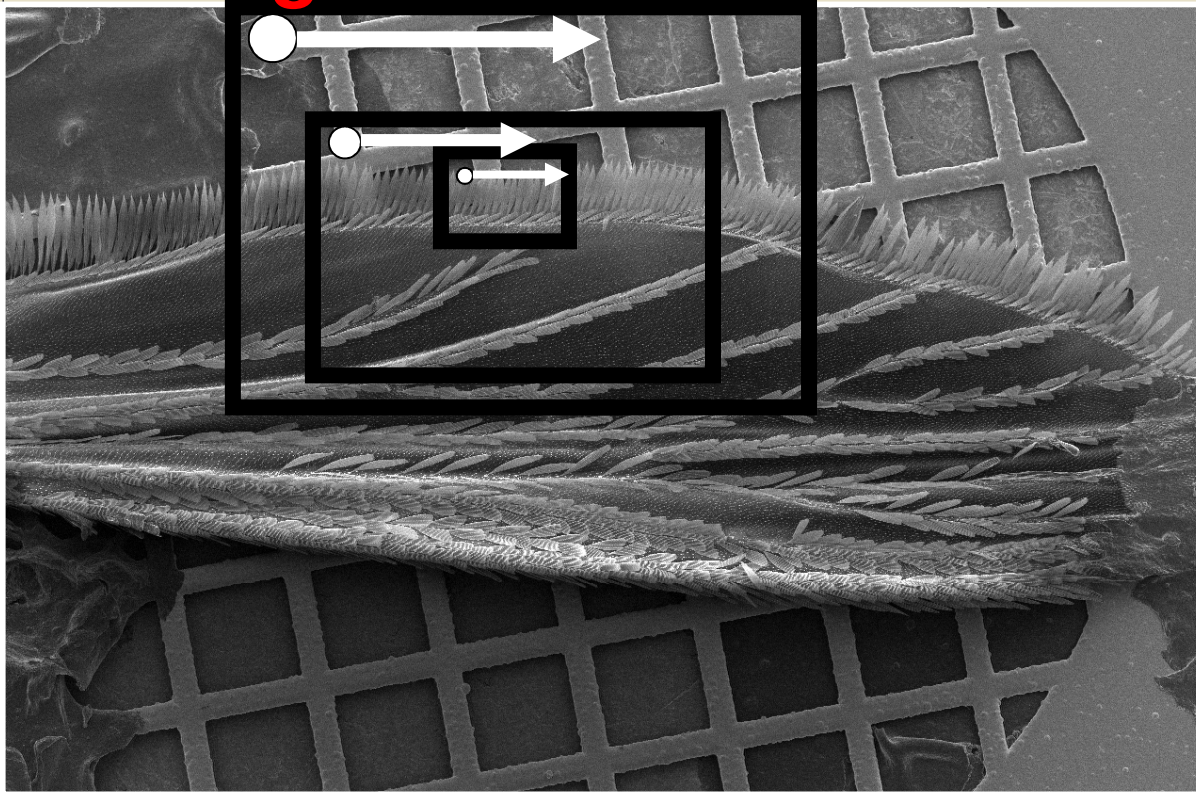


# Samples

- **Form:** any solid or liquid having a low vapour pressure ( $10^{-3}$  torr, or 0.13 pa)
- **Size:.** Samples as large as 4-8 cm.
- **Standard metallographic** polishing and etching techniques are adequate for electrically conducting materials.
- **Non-conducting materials** are generally coated with a thin layer of carbon, gold, or gold alloy. Samples must be electrically connected to the holder
- **Fine samples, such as powders,** can be dispersed on electrically conducting film, such as a silver point that has been thoroughly dried. Samples must be free from high vapor pressure liquids, such as water, organic cleaning solutions, and remnant oil-base films



# Magnification and resolution of SEM



Magnification increased by decreasing the area of scan on the specimen

A virtual microscope!

Resolution is a function of the spot size

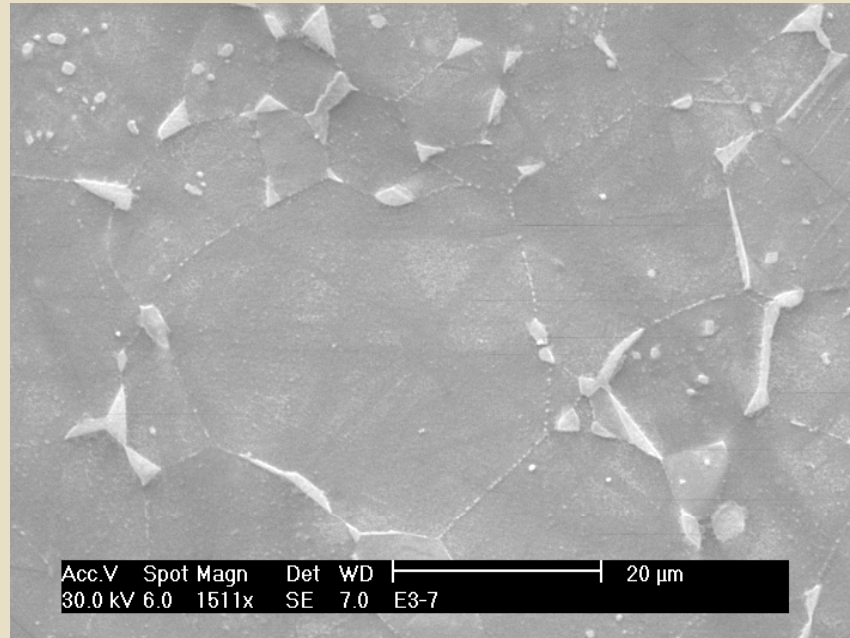
With a FEG gun the minimum spot size attainable  $\sim 1$  nm

Maximum useful magnification =  $0.2\text{mm}/1\text{nm} = 2,00,000$

Decreasing spot size – Improves resolution ✓

Intensity loss ✗

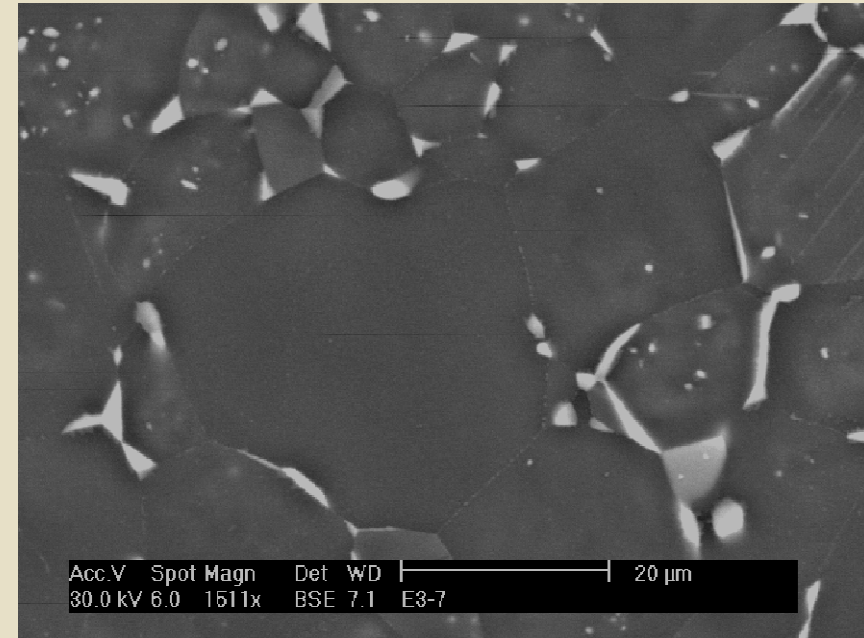
# Microstructure of a Ti-Ta-Nb alloy



SE imaging → Morphology

- Grain Size
- Area fraction of phases

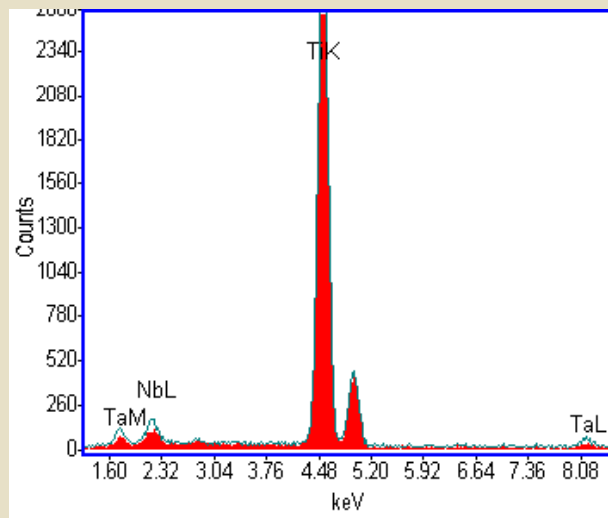
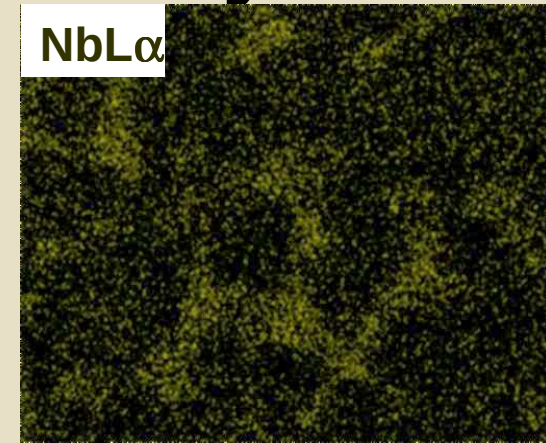
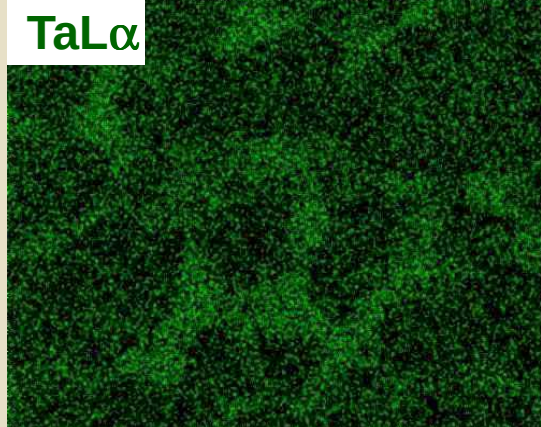
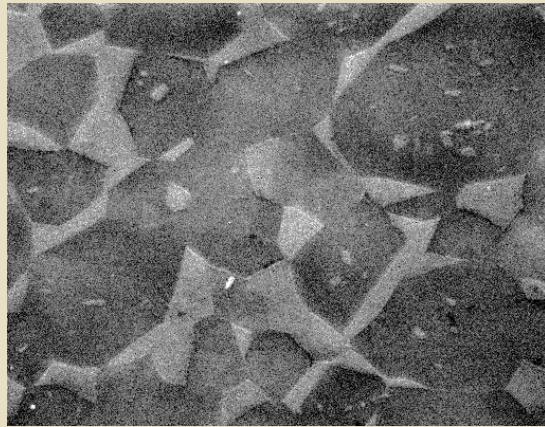
Chemical identity → EDS- X rays



BSE imaging → Bright – High Z  
Dark – Low Z



# Energy Dispersive X-ray analysis



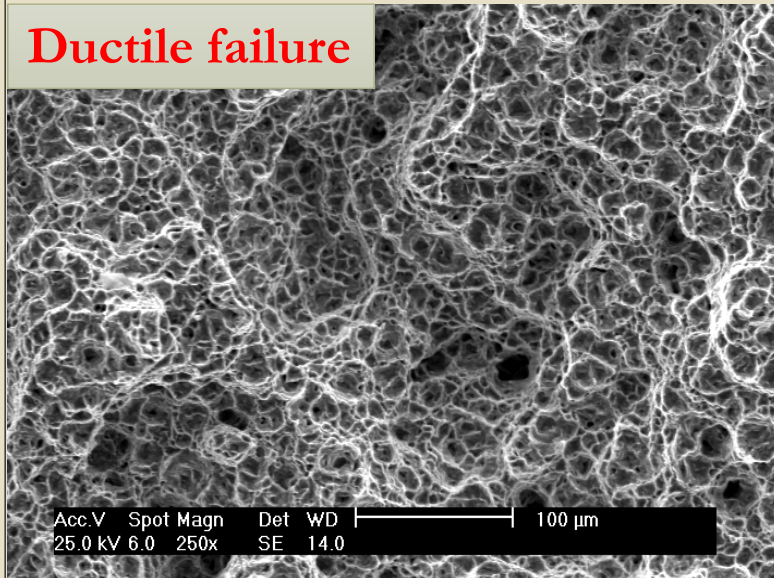
**Higher  
enrichment  
of Ta and Nb  
in  $\beta$  phase  
than in  $\alpha$   
phase**

- **Composition analysis from regions of a few microns size**
- **Spatial mapping of element distribution**

# Fractography

- No sample preparation required
- High depth of field  $\Rightarrow$  3D information

Ductile failure



Fatigue Striations



Intragranular

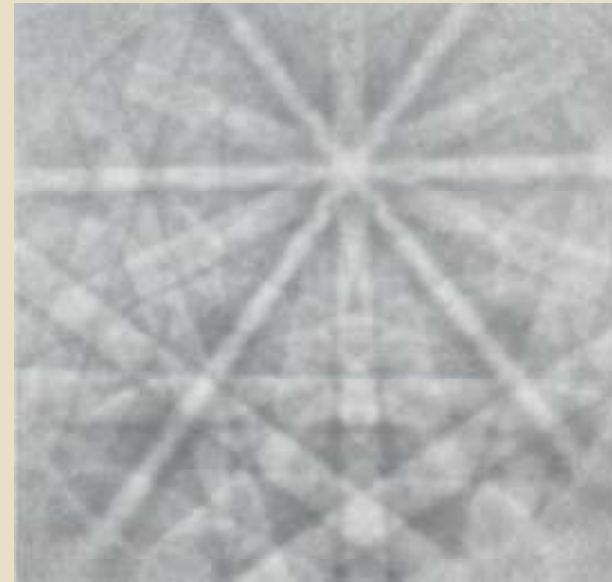
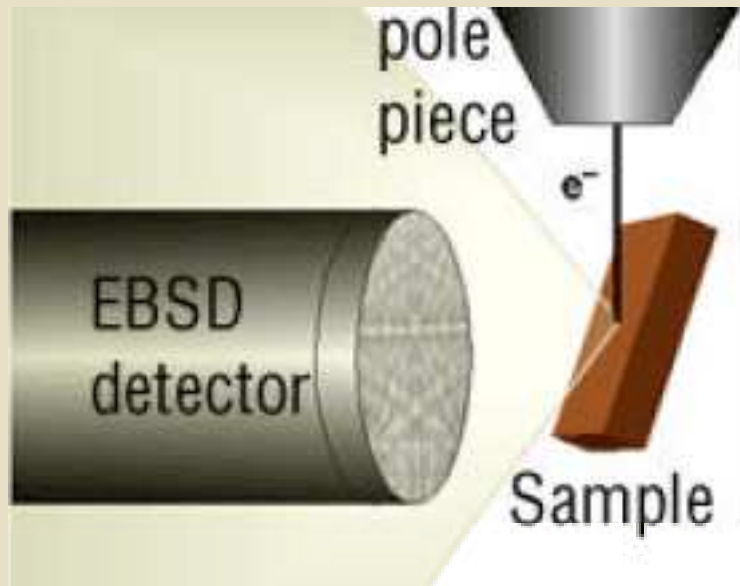


Intergranular



Identification of  
origin and  
mechanism of  
failure

# Electron Back Scattered Diffraction



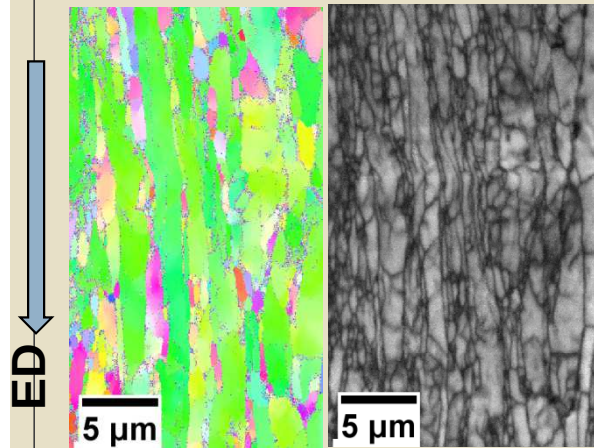
A typical EBSD pattern

- BSE undergo diffraction
- Diffracted electrons escaping from near surface produce Kikuchi bands imaged by film or a phosphor screen
- Symmetry of crystal lattice
- Width of bands are a measure of the interplanar spacing
- Angles between bands are related to the angles between planes in the lattice.



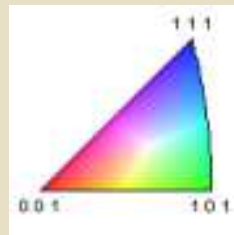
# Fabrication of 18Cr ODS Ferritic Steel clad tube

Extruded rod



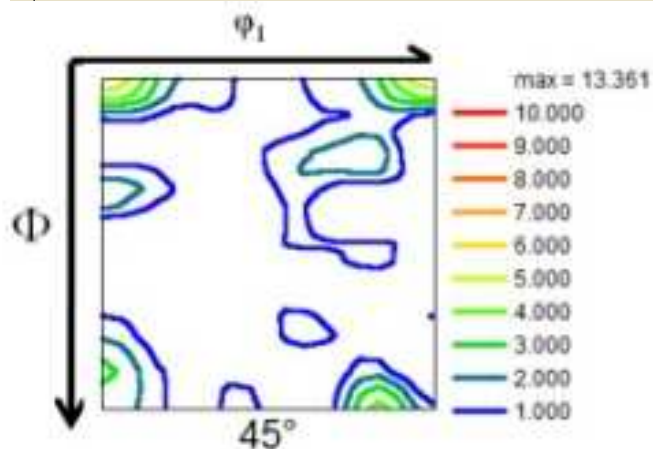
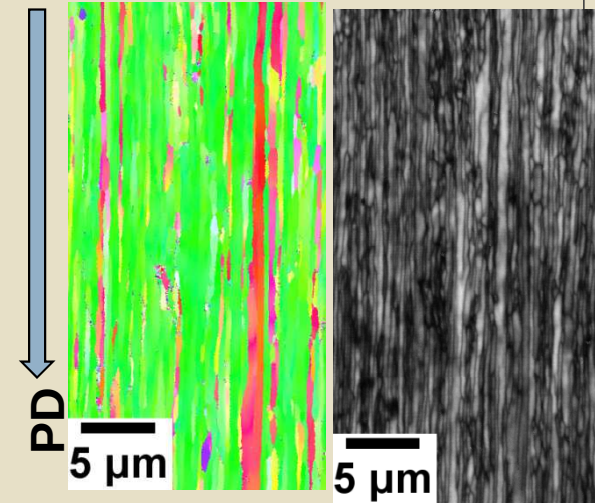
Pilgering

Double Int. annealing @  
1323 & 1423-1473K

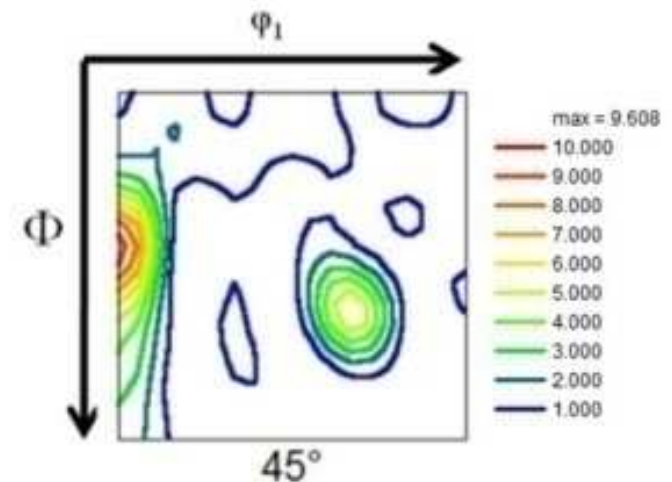


Elongated Bamboo like  
ferrite grains

Clad tube



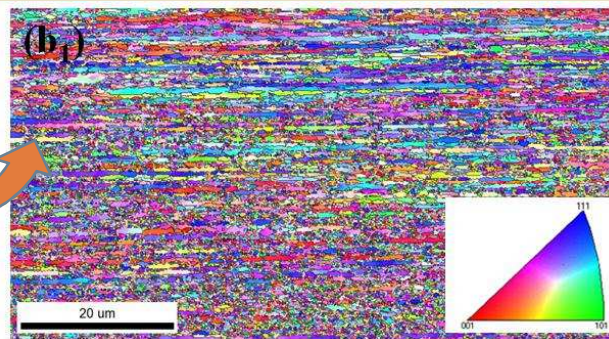
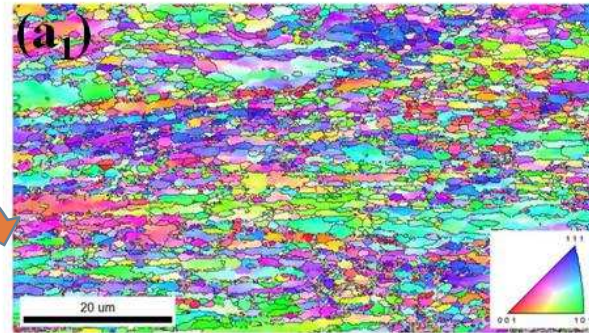
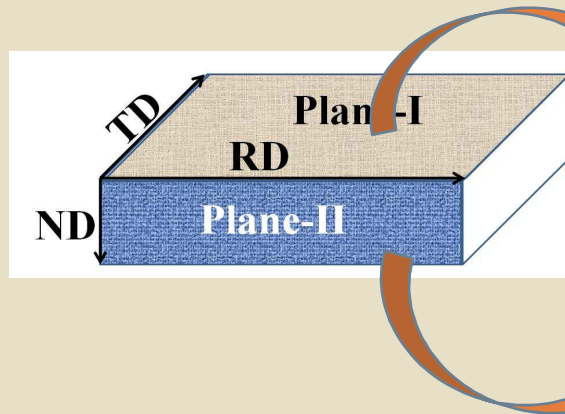
Very strong  $\alpha$  texture  
( $\langle 110 \rangle$  || ED/PD)



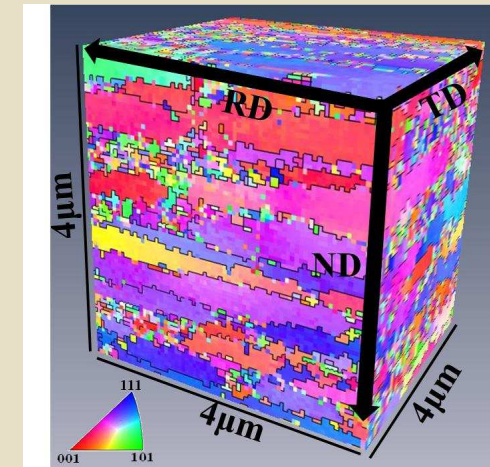


# Recrystallisation in Mod.9Cr-1Mo steel

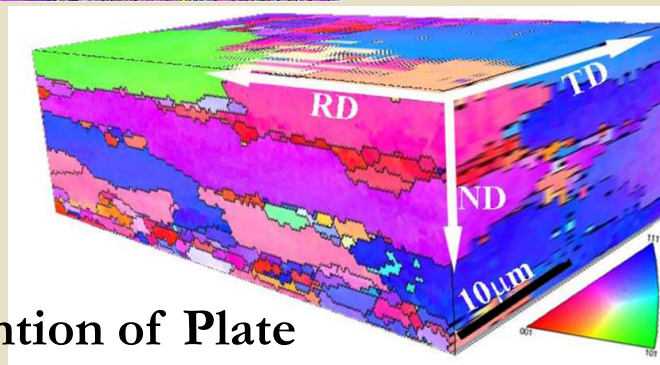
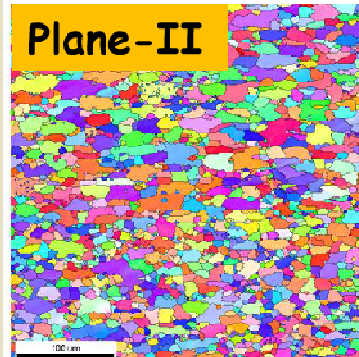
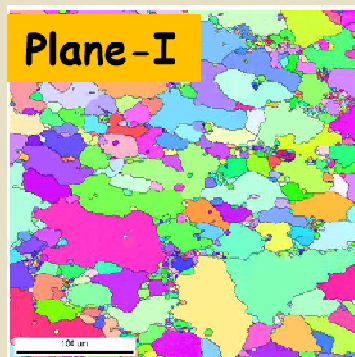
Cold Rolling (90%)



3D reconstruction



Recrystallisation @1023K



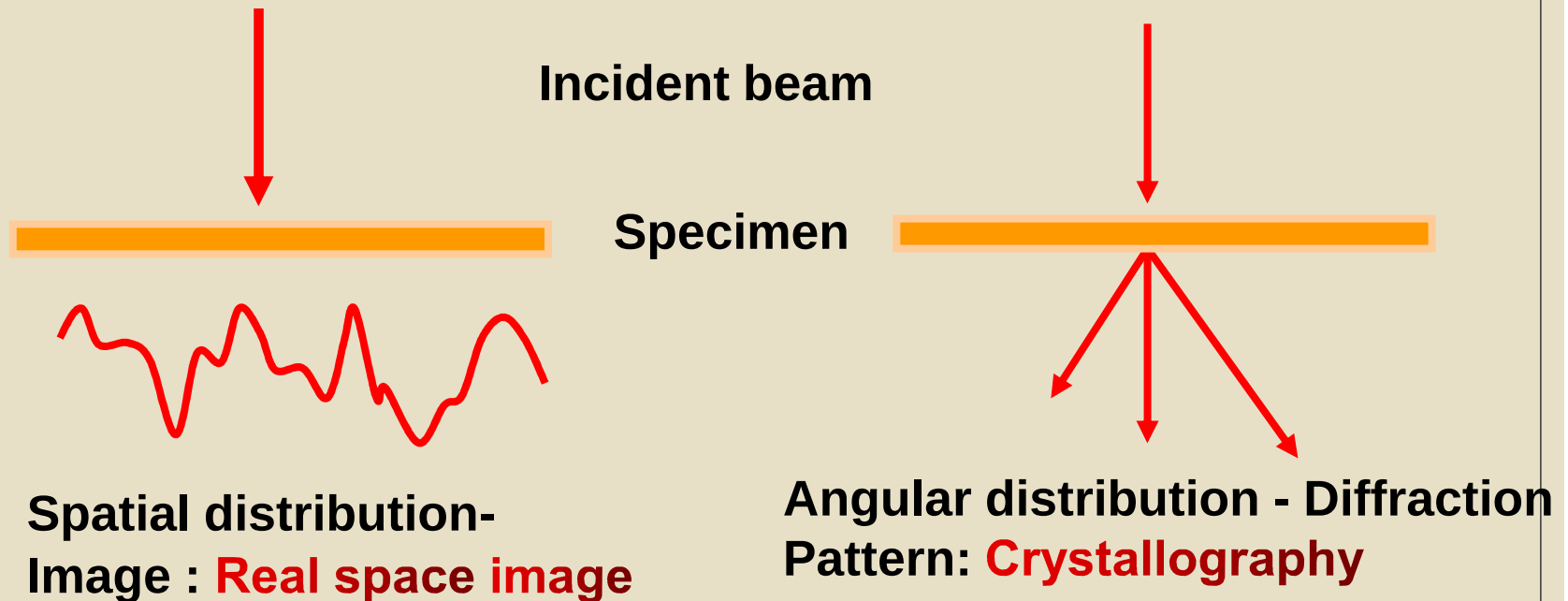
Retention of Plate like morphology

Partial recrystallisation

# Transmission Electron Microscopy

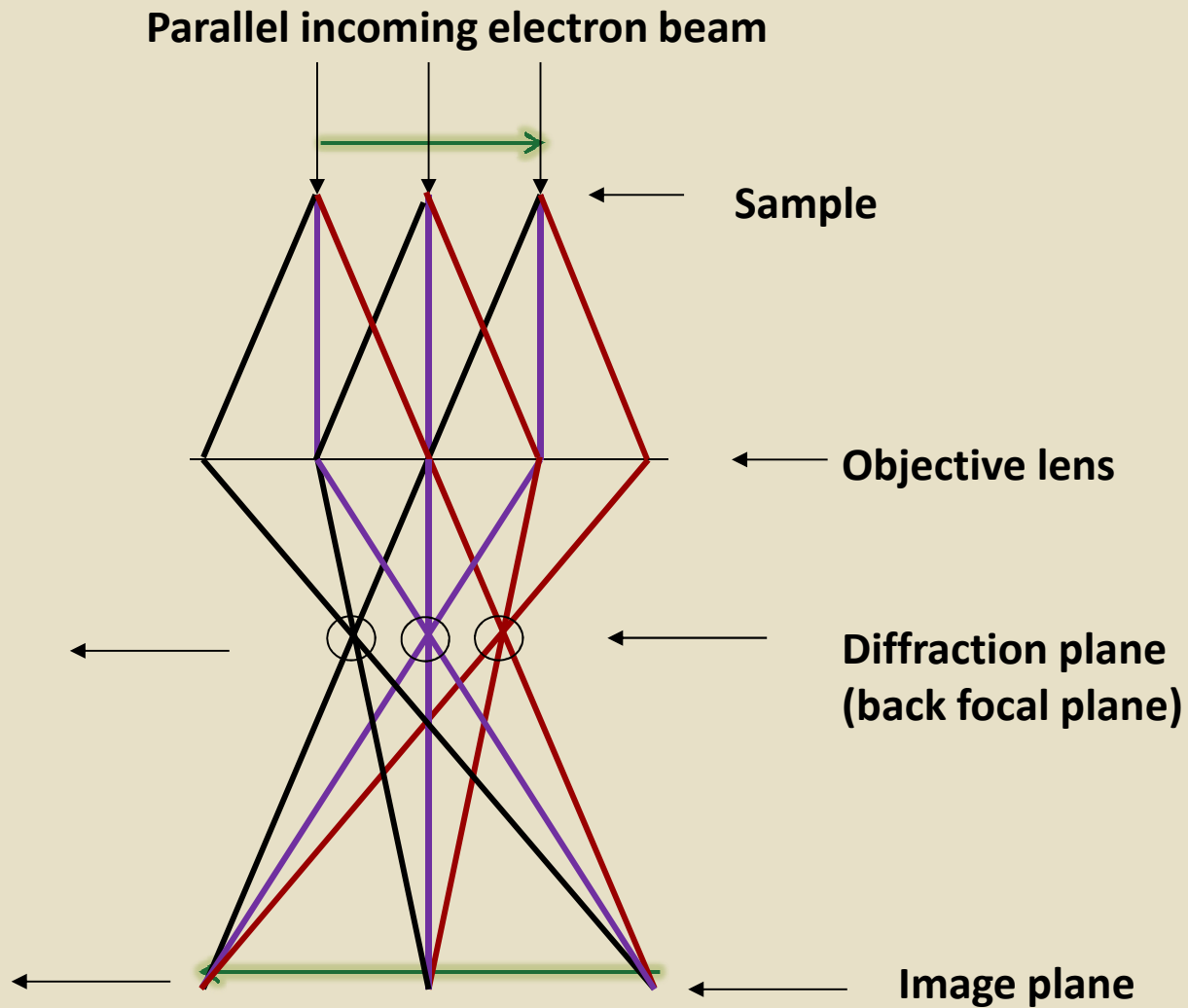
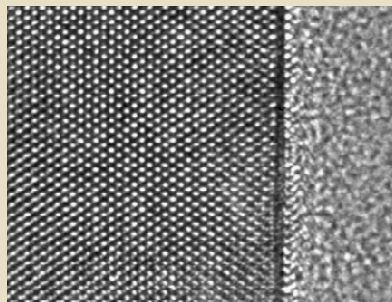
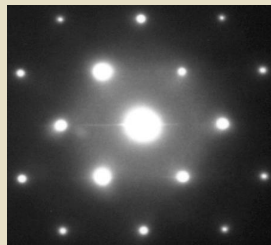
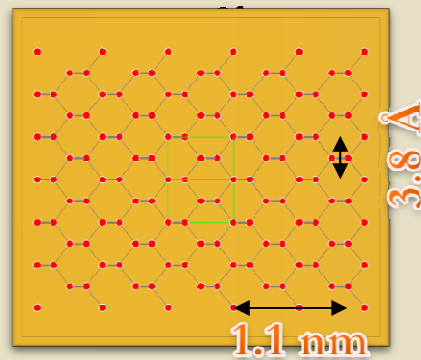
- Imaging of features at 1000 to 1000,000 X.
- Microstructural detail at resolution of  $<0.2\text{nm}$
- Qualitative and quantitative elemental analysis of microstructural features as small as 10 nm
- Crystal structure and orientation determination of microstructural features as small as 10 nm
- Lattice imaging of crystals with inter-planar spacing  $> 0.12\text{ nm}$
- Characterization of microstructure at very high magnification - metals, ceramics, geological materials, polymers, and biological materials
- Identification (composition and crystal structure) of inorganic phases, precipitates, and contaminants

# Diffraction and Imaging in TEM



- Amplitude contrast
  - Mass-thickness
  - Diffraction
- Phase contrast

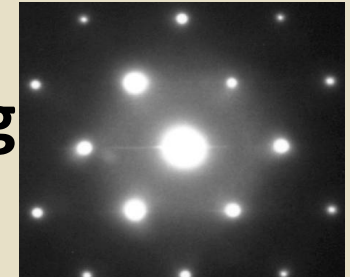
# Simplified ray diagram



# Electron diffraction

- Elastic scattered electrons

- Only the direction of electron is changing (Bragg scattering)



- Elastic scattering is due to Coulomb interaction between the incident electrons and the electric charge of electron cloud and nucleus (Rutherford scattering).
- The elastic scattering is due to the average position of the atoms in the lattice
- Reflections satisfying Braggs law:  $2d\sin\theta = n\lambda$

**Electrons interact 100-1000 times stronger with matter than X-rays**

- more absorption (need thin samples)

- can detect weak reflections not observed with X-rays



# Electron Diffraction in TEM

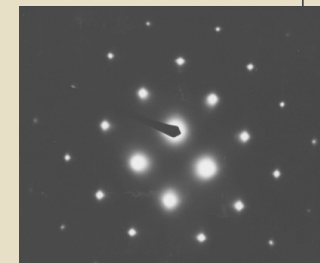
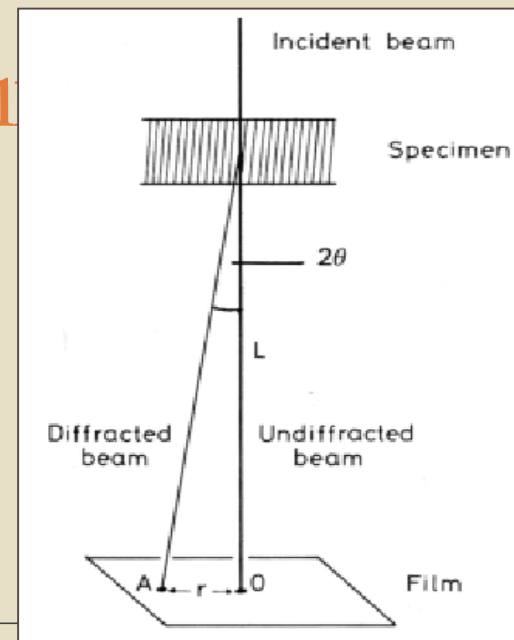
- Consequences of small  $\lambda$ 
  - Electron wave length at 200 kV = 2.508 pm
  - $1/\lambda \sim 400 \text{ nm}^{-1}$
  - $|g_{hkl}| = 1/d_{hkl} = 5 \text{ nm}^{-1}$  ( typically)
- Very small Bragg angle ( $\sim 1^\circ$ )
  - **Diffraction planes are nearly parallel to the primary beam**

From geometry, noting  $\theta$  small  
(Bragg law:  $2d\theta = \lambda$ )

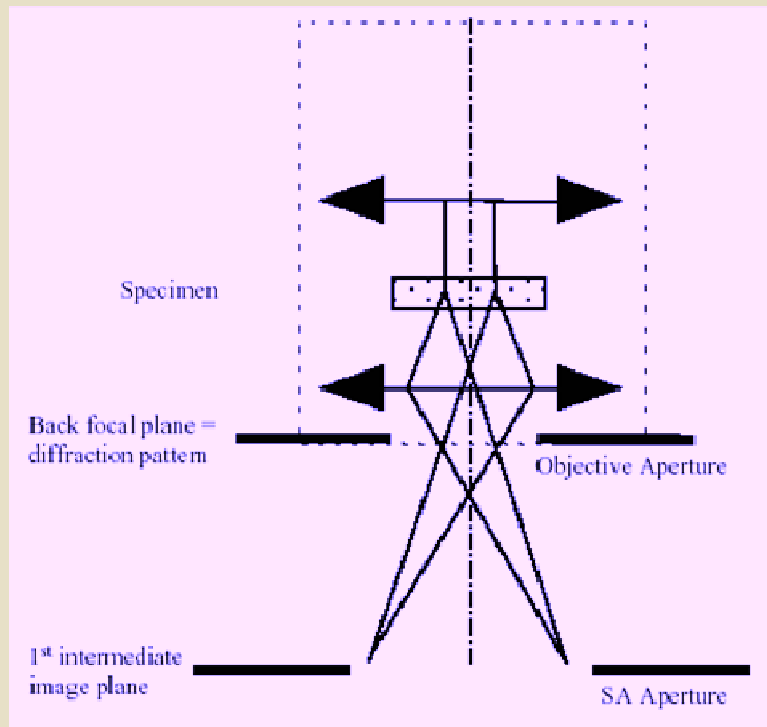
$$r = L 2\theta = L \lambda / d_{hkl}$$

$$r d_{hkl} = L \lambda$$

$L \lambda$ : camera constant

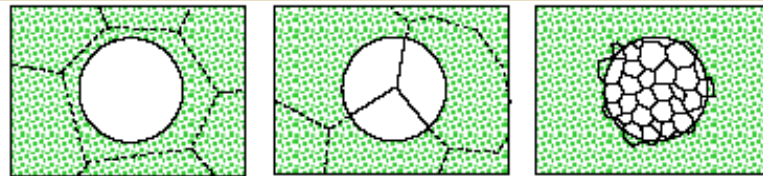


# Selected area electron diffraction



- A specimen which is crystalline will deflect incoming electrons in specific directions.
- For a parallel incident beam all electrons diffracted over the same angle will be focused at the same position in the BFP of objective lens.
- Area selected by aperture placed in image plane
- Virtual aperture created in specimen plane – demagnified !

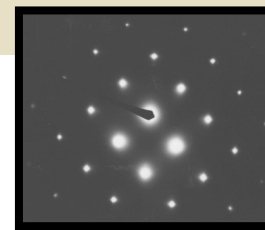
## Changes in diffraction pattern – spots to rings



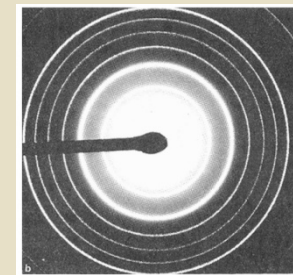
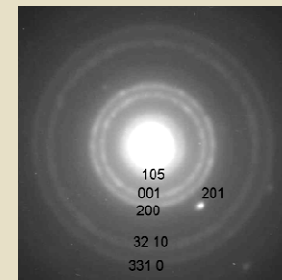
(a)

(b)

(c)



Single crystal



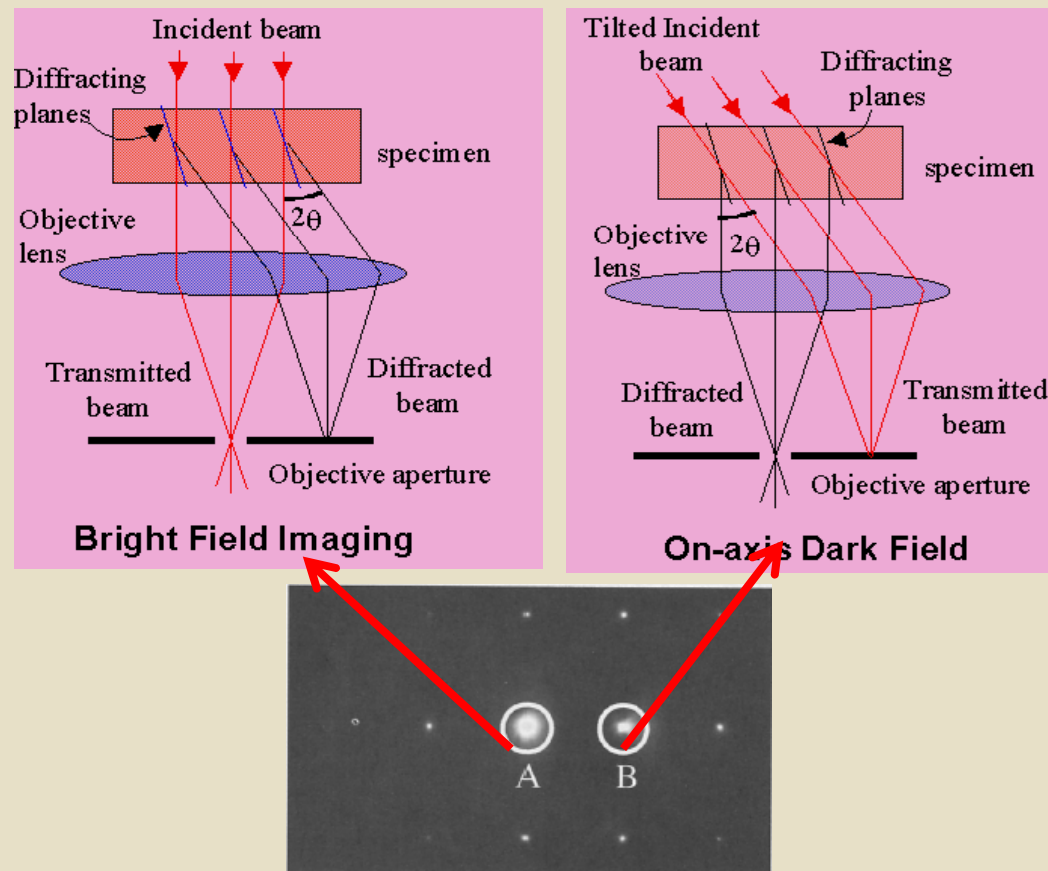
polycrystal



Amorphous

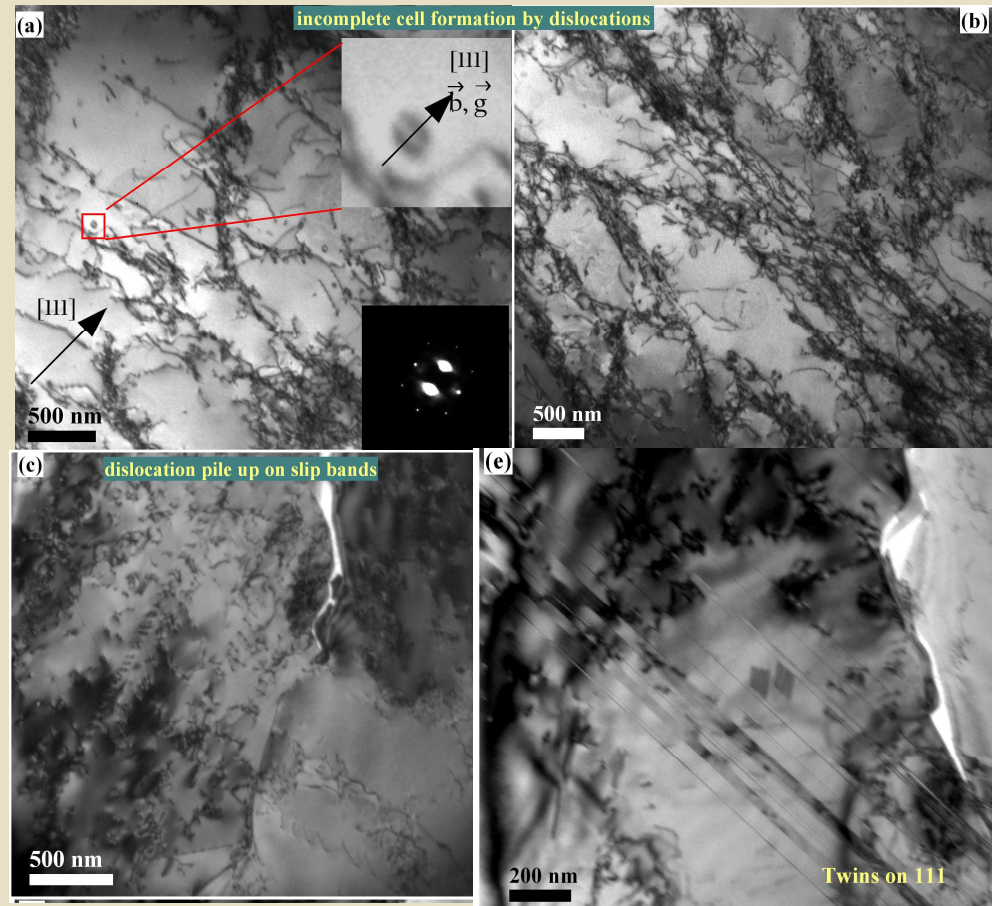
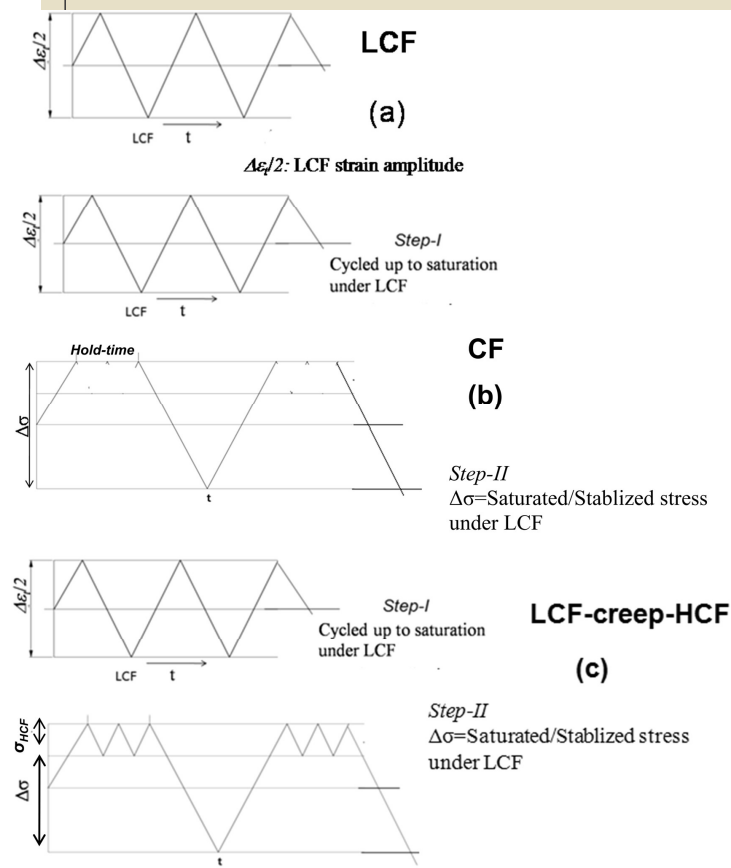
# Bright Field / Dark Field modes

- Selective filtering of scattering that occurs at Bragg angles
- Dark-field image contains specific orientation information not just general scattering information as is the case for mass-thickness contrast



# DEFECT STRUCTURE EVOLUTION IN SS316LN UNDER CYCLIC LOADING

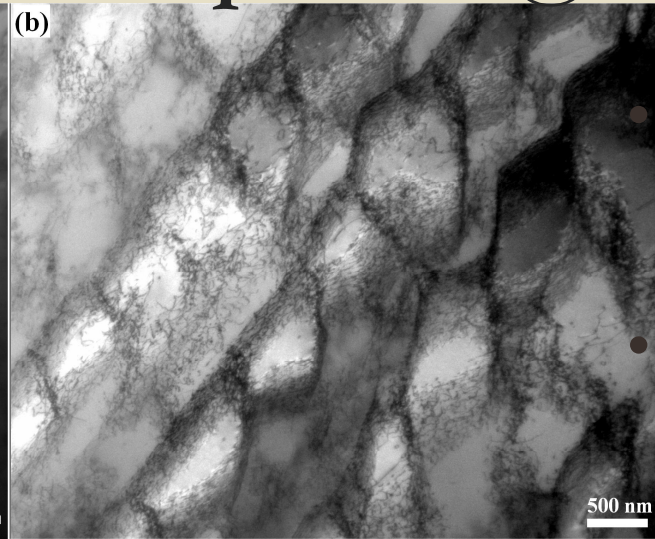
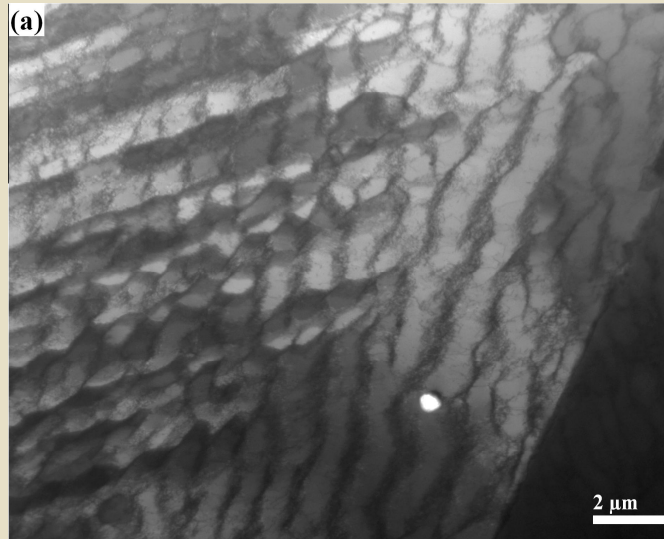
Strain amplitude ( $\Delta\epsilon/2$ ) of  $\pm 0.25\%$ ,  $T = 923\text{K}$



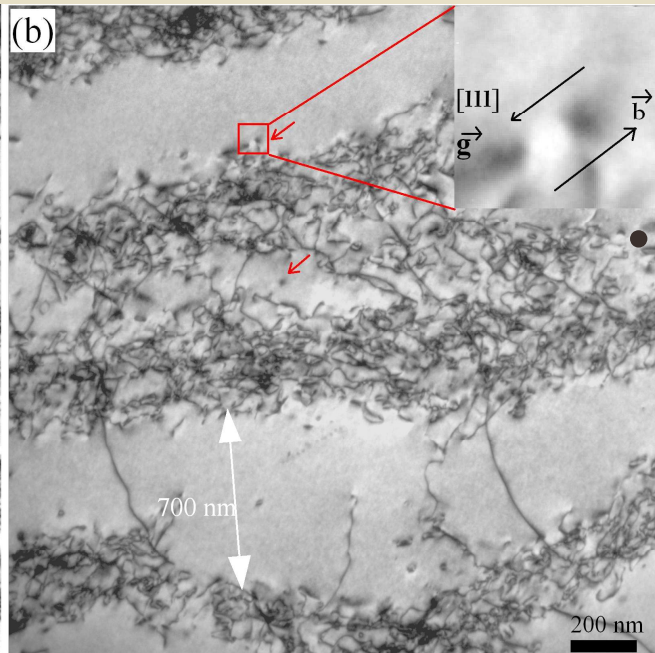
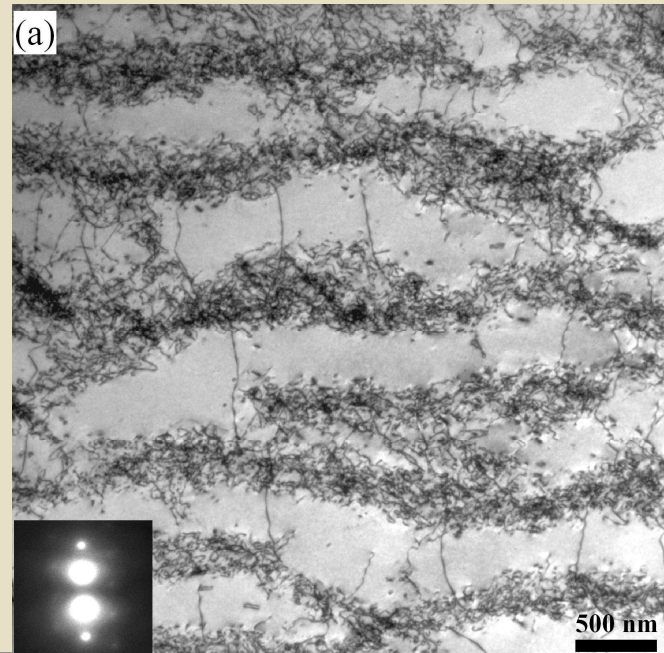
- Slip bands with dislocation pile ups
- Dislocation loops
- Fine Deformation twins (111)



# SS316LN- LCF- Creep-HCF@ 923K

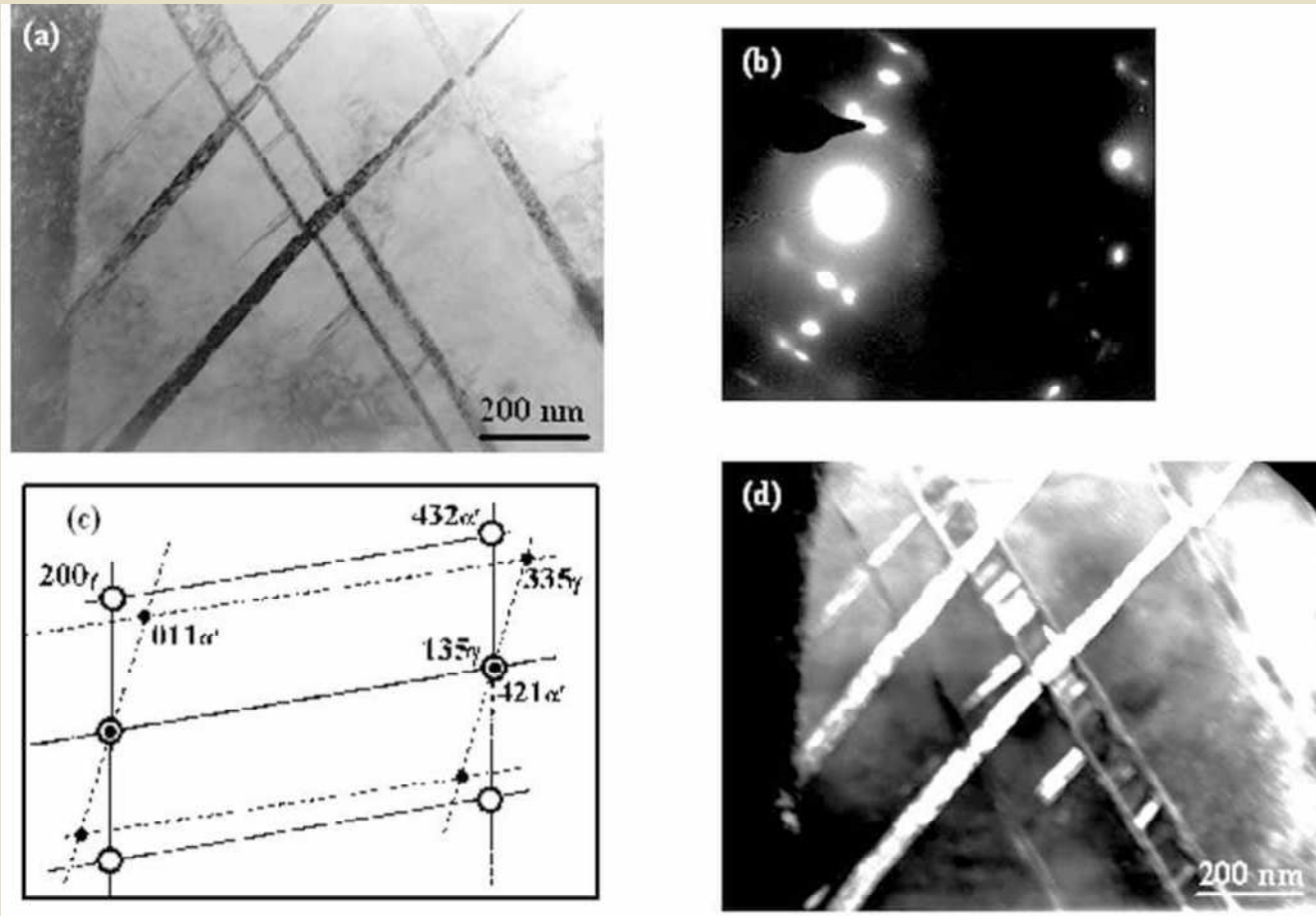


- Well defined dislocation cells
- Dislocation vein structure with high density dislocation walls
- Point defects like Frank edge-on loops





# Ti modified 316 stainless steel bellow



- Partial conversion of  $\gamma \rightarrow \alpha'$
- Strain induced martensite formed during fabrication
- Nucleation centres for  $\alpha'$

# Summary

Characterization is essential for

- Understanding deformation mechanism
- Development of materials and processes
- Life assessment/prediction

**Several tools available to study various aspects of characteristics of materials from component to atomic scale, with technological advancements!**

- Morphology
- Crystal Structure
- Texture
- Composition

THANK YOU